

Silicon Detectors - I

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Department of Physics





**Micro- and Nanotechnology Laboratory (MiNaLab), 5000 m², OSLO
(April-2004)**



MiNa-lab



...inside



MiNa-lab (Oslo, Aug-2002)



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Staff at MiNa

PhD students

Giovanni Alfieri

Marc Avice

Jan H. Bleka

Thomas Moe Børseth

Ingelin Clausen

K-M. Johansen

Matthieu Lacolle

Jeyanthinath Mayandi

Mads Mikelsen

Ramon Schifano

Lasse Vines

2 New PhD students

Post-doctors/Researchers

Jens S. Christensen

Ulrike Grossner

Philip Y.Y. Kan

Eduoard V. Monakhov

Leonid Murin

Ioana Pintilie

Alexander Ulyashin

2 New Post-docs

~6 MSc students, Visitors

'Permanent'

Viktor Bobal, engineer

Terje G. Finstad, prof

Liv Furuberg, ass. prof (adj)

A. B. Hanneborg, ass. prof (adj)

Andrej Yu. Kuznetsov, ass. prof

Thomas Martinsen, engineer

Ola H. Sveen, ass. prof

Bengt G. Svensson, prof

Aasmund Sudbø, prof (UNIK)

David I. Wormald, eng. (30 %)

Jan-2006

Outline



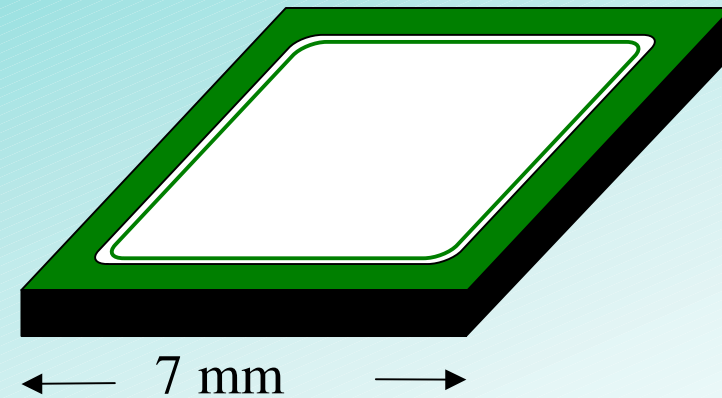
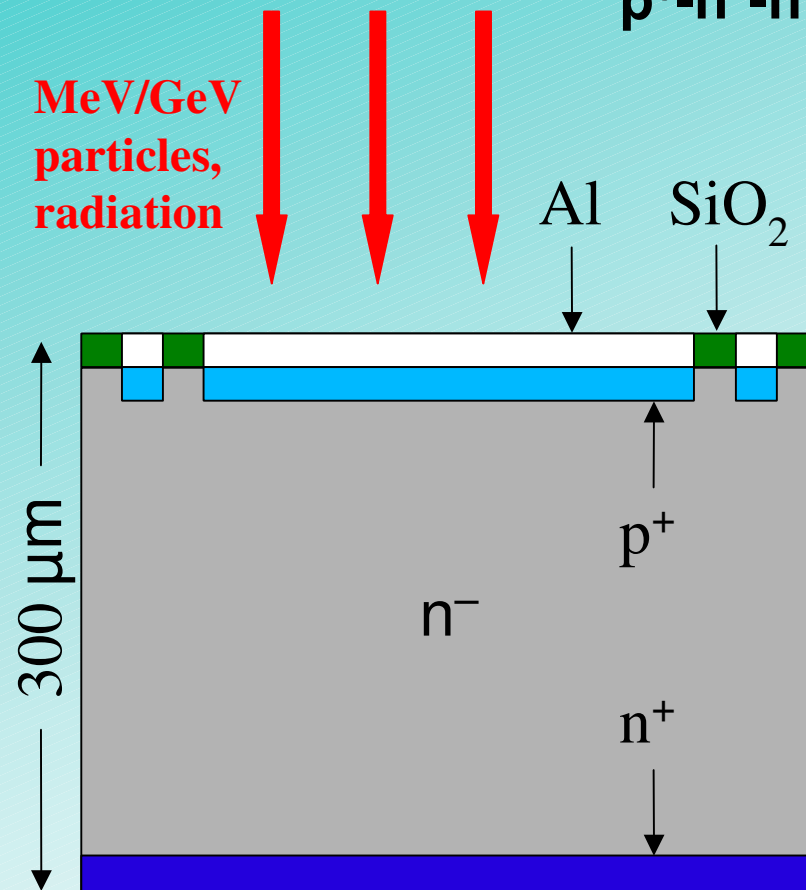
- ⌘ Some fundamentals about materials, device physics and processing related to Si detectors
- ⌘ Radiation tolerance of Si detectors, limiting factors
- ⌘ Defect/impurity engineering of Si detectors
- ⌘ New detector structures

'Baby detector'



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p⁺-n⁻-n⁺ Si detectors



$$n^- \sim (2-5) \times 10^{12} \text{ cm}^{-3}$$

$$\text{Cf } n_i \sim 1.5 \times 10^{10} \text{ cm}^{-3} @ \text{ RT}$$

p⁺, n⁺ by ion implantation



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Department of Physics/Physical Electronics

MeV ion accelerator at UiO/MiNa-lab

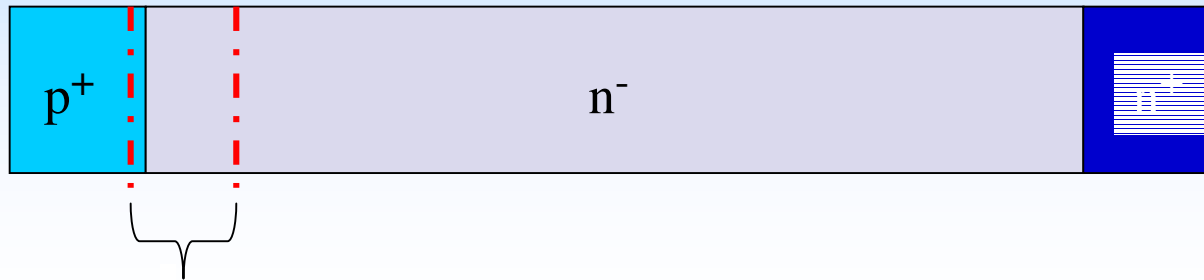
Ion implantation and RBS-analysis



National Electrostatics Corporation, 1 MV terminal voltage

Basic considerations

(zero bias)



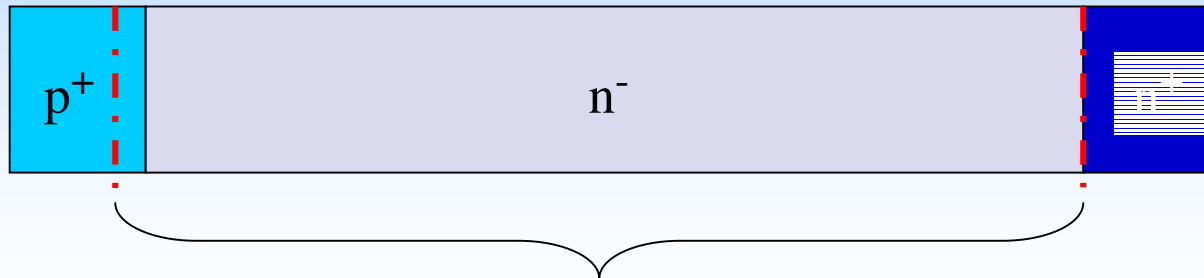
$$W = \left(2\epsilon V_0 (N_a + N_d) / q(N_a + N_d) \right)^{1/2} - \text{depletion region}$$

$$N_a \gg N_d \longrightarrow W = \left(2\epsilon V_0 / (qN_d) \right)^{1/2} \approx 20 \mu\text{m}$$

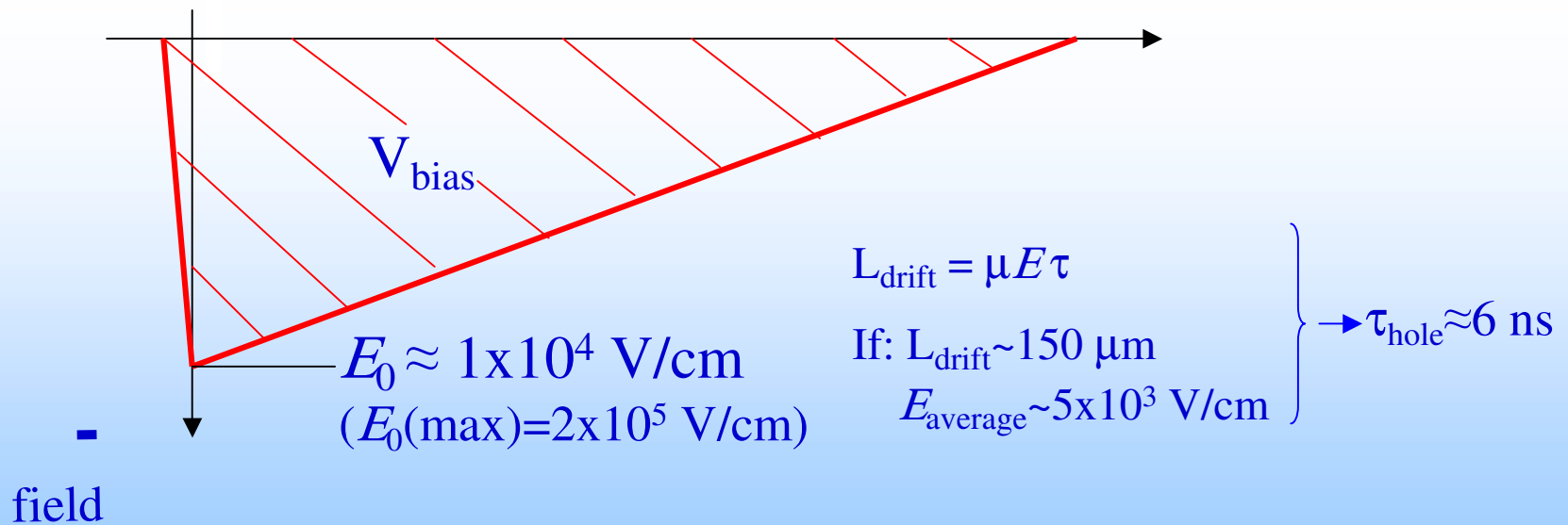
$$L_D = \left(\epsilon kT / q^2 N_d \right)^{1/2} \approx 3 \mu\text{m} - \text{Debye length at RT}$$

Basic considerations

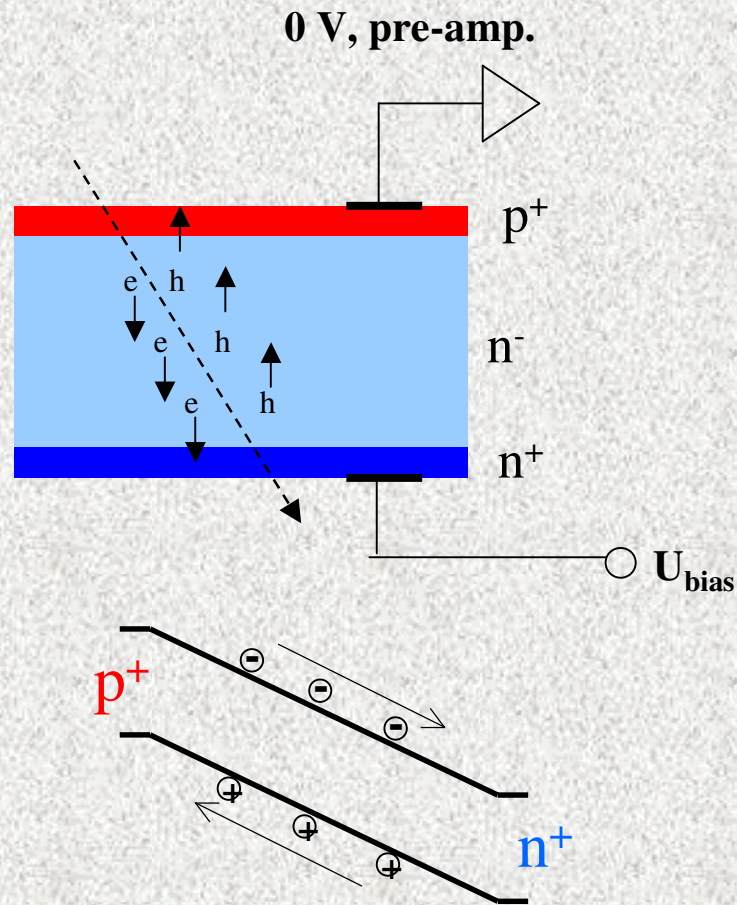
(full (over) depletion)



$W \approx 300 \mu\text{m} \rightarrow V_{\text{reverse bias}} \approx 135 \text{ V}$



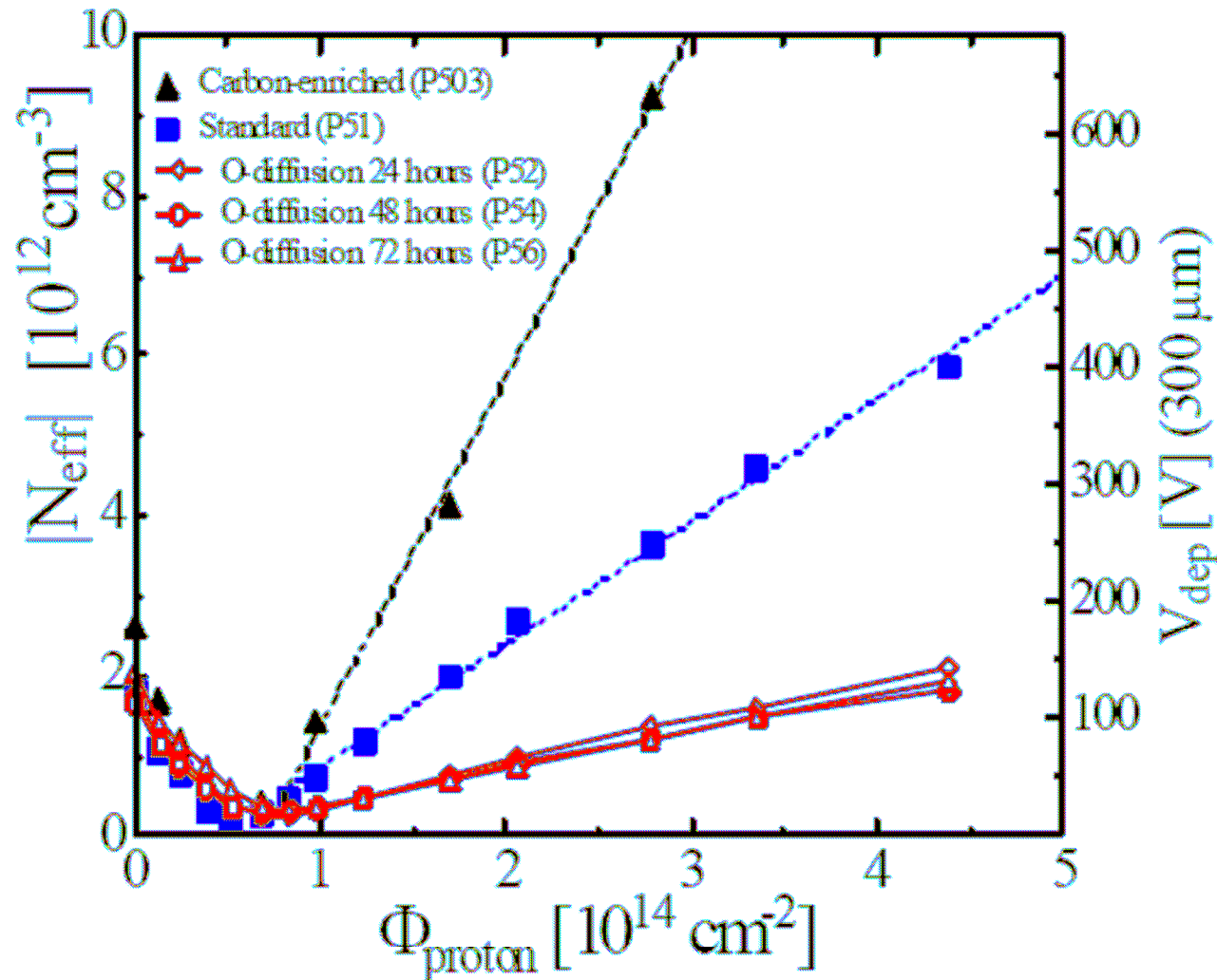
Silicon particle detectors



Advantages:

- high signal-to-noise ratio
- fast direct charge readout
- high spatial resolution

Impurity engineering of high-purity Si-detectors



G.Lindström et al. (The RD48 Collaboration); "Developments for Radiation Hard Silicon Detectors by Defect Engineering - Results by the CERN RD48 (ROSE) Collaboration –"NIM A 465/1 (2001) 60-69.

Signal formation

$$Q = \int_{t=0}^t I dt = q \int_{t=0}^t \vec{v} \vec{E}_w dt = q \int_{\vec{r}_0}^{\vec{r}(t)} \vec{E}_w d\vec{r}$$

$$Q = q[U_w(\vec{r}) - U_w(\vec{r}_0)]$$

$$Q_{e-h} = Q_e + Q_h$$

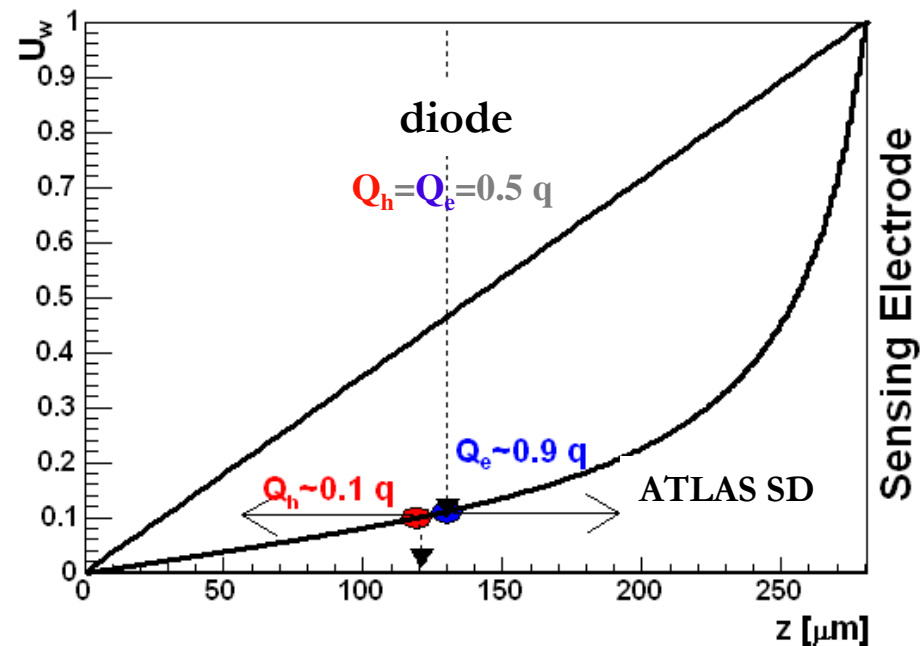
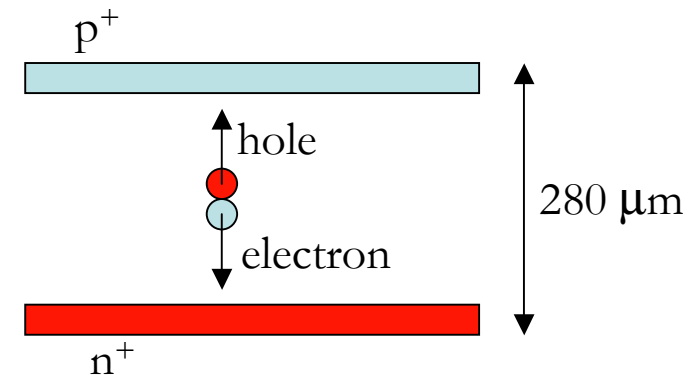
Contribution of drifting carriers to the total induced charge depends on ΔU_w !

Simple in diodes and complicated in segmented devices!

For track:

$$Q_e/(Q_e + Q_h) = 19\%$$

in ATLAS strip detector



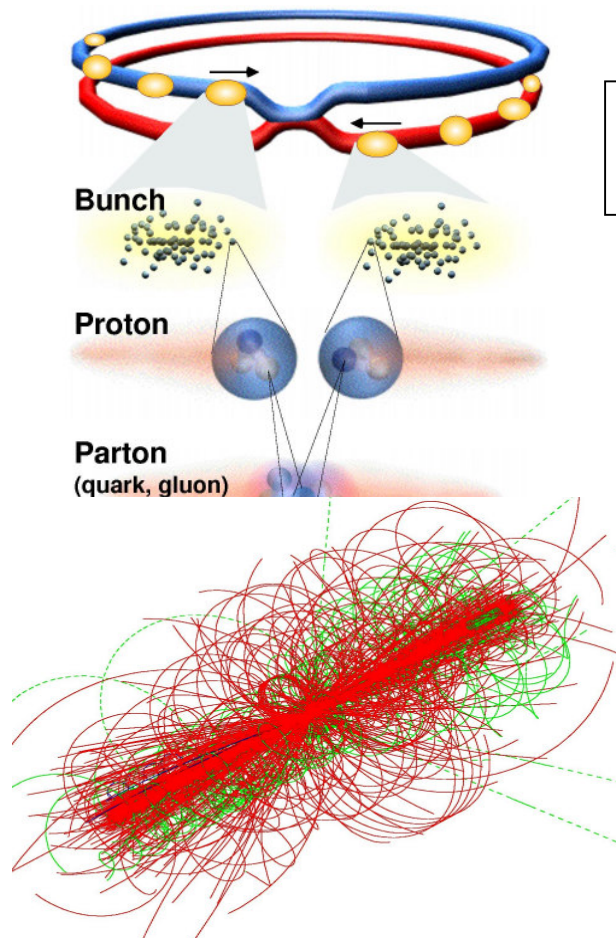
Outline



- ⌘ Some fundamentals about materials, device physics and processing related to Si detectors
- ⌘ Radiation tolerance of Si detectors, challenges and limiting factors
- ⌘ Defect/impurity engineering of Si detectors
- ⌘ New detector structures

Silicon Detectors: Favorite Choice for Particle Tracking

Example: Large Hadron Collider LHC, start 2007



LHC properties

- Proton-proton collider, 2 x 7 TeV
Luminosity: 10^{34}
- Bunch crossing: every 25 nsec, Rate: 40 MHz
event rate: 10^9 /sec (23 interactions per bunch crossing)
- Annual operational period: 10^7 sec
- Expected total op. period: 10 years

Experimental requests

Detector properties

- | | | |
|--|---|--|
| Reliable detection of mips | → | S/N ≈ 10 |
| High event rate | → | time + position resolution: |
| high track accuracy | → | ~ 10 ns and ~ 10 μm |
| Complex detector design | → | low voltage operation in normal ambients |
| Intense radiation field during 10 years | → | Radiation tolerance up to 10^{15} hadrons/cm² |
| Feasibility, e.g. 200 m ² for CMS | → | large scale availability
known technology, low cost |

! Silicon Detectors meet all Requirements !

Main motivations for R&D on Radiation Tolerant Detectors: Super - LHC

- **LHC upgrade**

⇒ **LHC (2007)**, $L = 10^{34} \text{cm}^{-2}\text{s}^{-1}$

10 years
→
500 fb⁻¹

$$\phi(r=4\text{cm}) \sim 3 \cdot 10^{15} \text{cm}^{-2}$$

CERN-RD48

⇒ **Super-LHC (2015 ?)**, $L = 10^{35} \text{cm}^{-2}\text{s}^{-1}$

5 years
→
2500 fb⁻¹

$$\phi(r=4\text{cm}) \sim 1.6 \cdot 10^{16} \text{cm}^{-2}$$

CERN-RD50

× 5

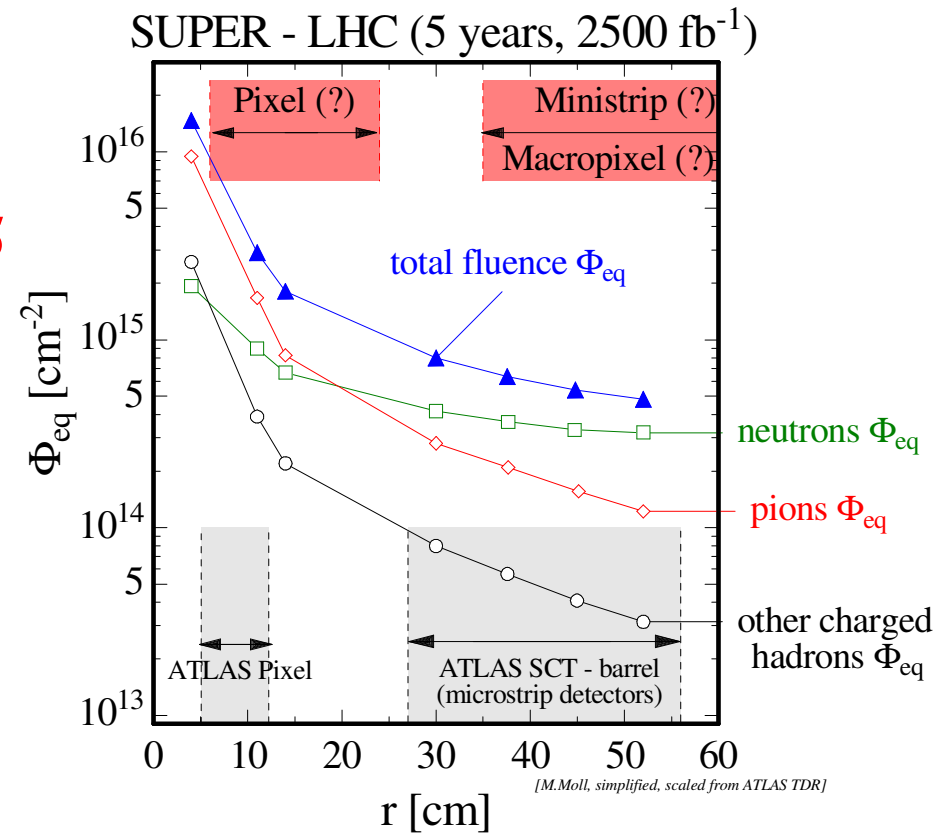
- **LHC (Replacement of components)**

e.g. - LHCb Velo detectors (~2010)

- ATLAS Pixel B-layer (~2012)

- **Linear collider experiments (generic R&D)**

Deep understanding of radiation damage will be fruitful for linear collider experiments where high doses of e, γ will play a significant role.



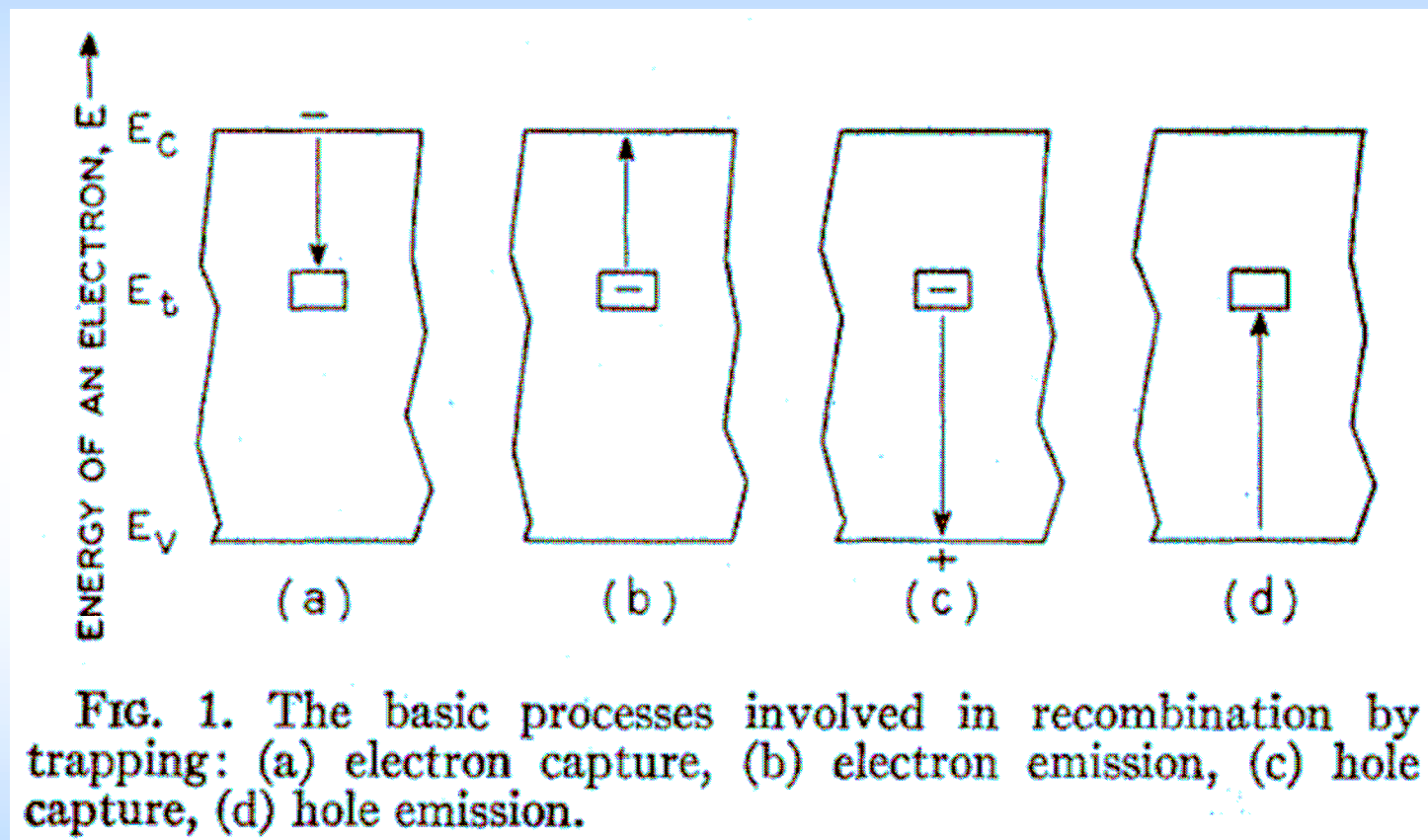
Radiation Damage in Silicon Detectors

Two types of radiation damage in detector structures:

- Bulk (Crystal) damage due to Non Ionizing Energy Loss (NIEL)
 - displacement damage, built up of crystal defects –
 - I. Increase of **leakage current** (increase of shot noise, thermal runaway)
 - II. Change of **effective doping concentration**
(higher depletion voltage, under- depletion)
 - III. Increase of **charge carrier trapping** (loss of charge)
- Surface damage due to Ionizing Energy Loss (IEL)
 - accumulation of charge in the oxide (SiO_2) and Si/ SiO_2 interface –
affects: **interstrip capacitance** (noise factor), **breakdown behavior**, ...

! Signal/noise ratio = most important quantity !

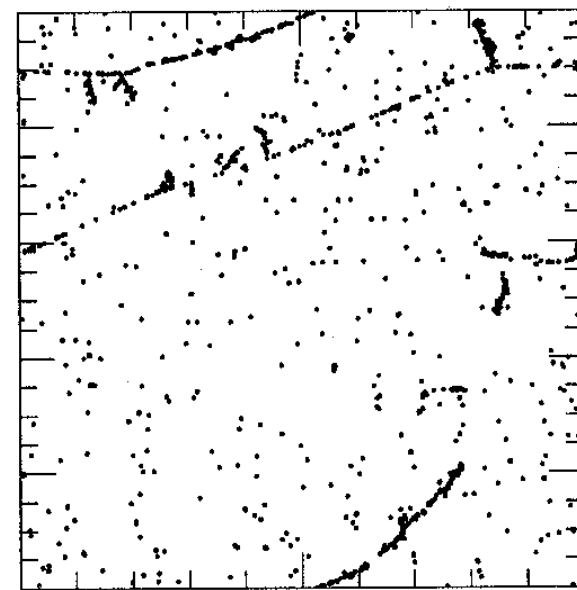
Trapping and recombination of carriers



Deterioration of Detector Properties by displacement damage NIEL

Point defects
+ clusters

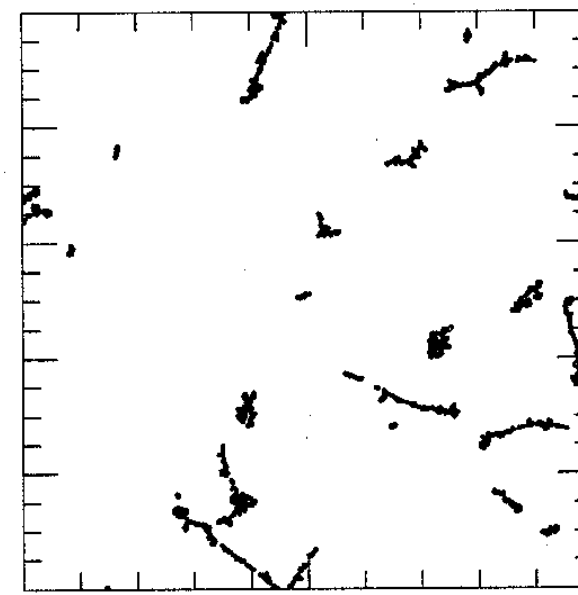
4145 vacancies



0 0.5 1
x (μm)

Dominated
by clusters

8870 vacancies



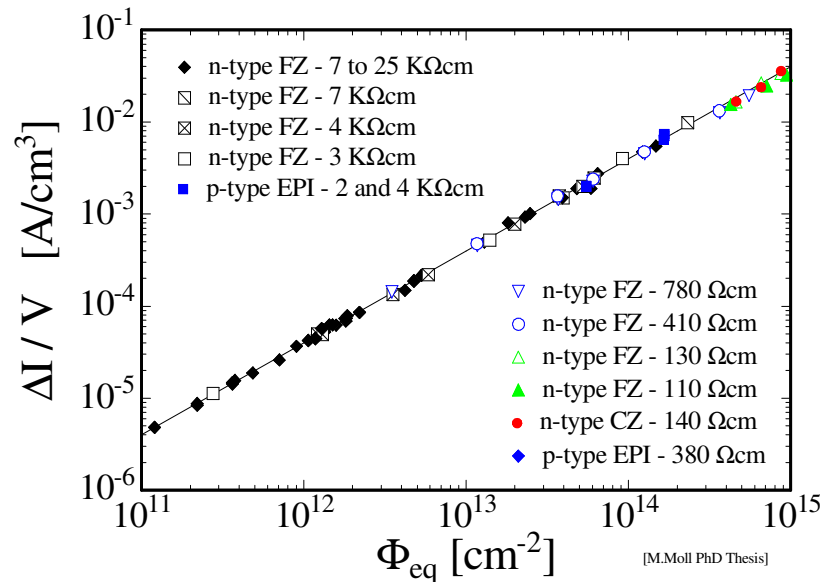
0 0.5 1
x (μm)

Damage effects generally $\sim$ NIEL, however differences between proton & neutron damage important for defect generation in silicon bulk

Radiation Damage – Leakage current

Increase of Leakage Current

.... with particle fluence:



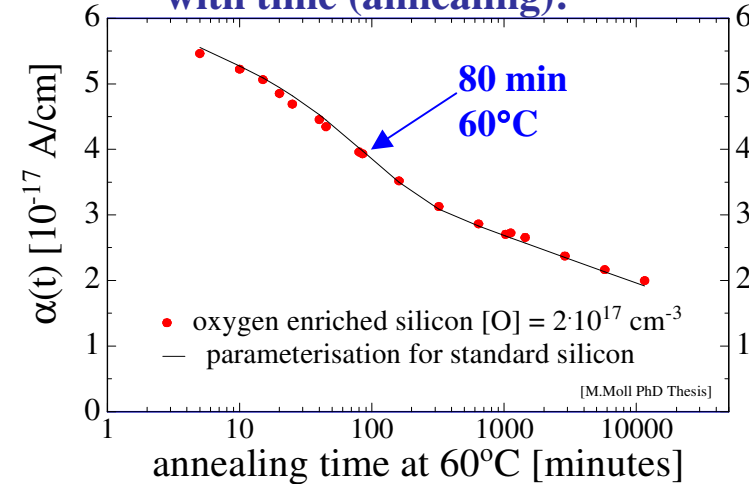
- Damage parameter α (slope in figure)

$$\alpha = \frac{\Delta I}{V \cdot \Phi_{eq}}$$

Leakage current
per unit volume
and particle fluence

- α is constant over several orders of fluence and independent of impurity concentration in Si
⇒ can be used for fluence measurement

with time (annealing):



- Leakage current decreasing in time (depending on temperature)
- Strong temperature dependence:

$$I \propto \exp\left(-\frac{E_g}{2k_B T}\right)$$

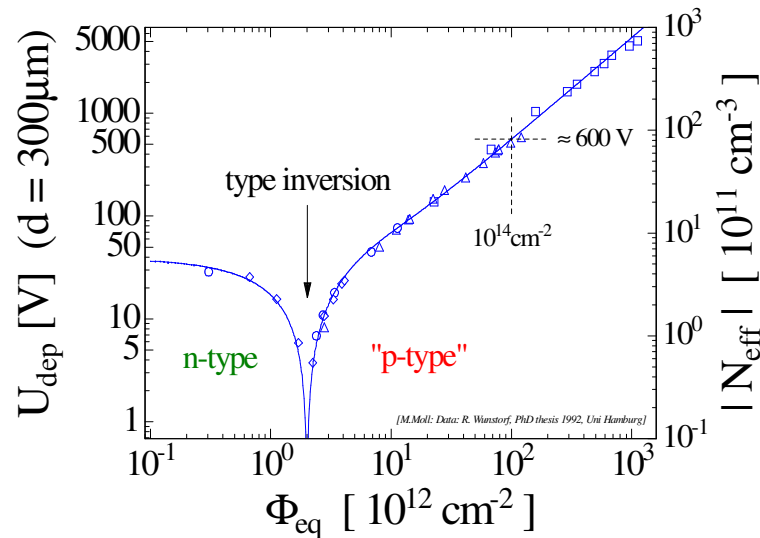
Consequence:

Cool detectors during operation!
Example: $I(-10^\circ\text{C}) \sim 1/16 I(20^\circ\text{C})$

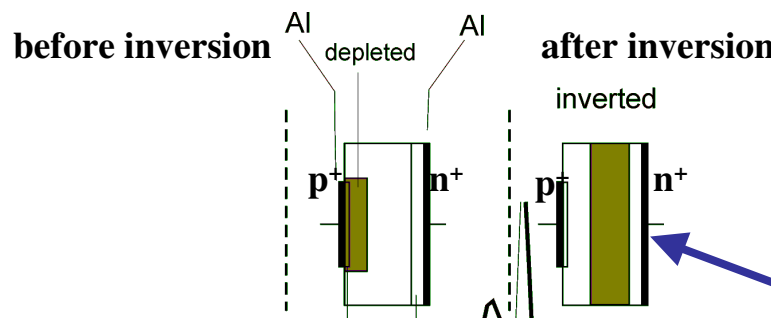
Radiation Damage – Effective doping concentration

Change of Depletion Voltage V_{dep} (N_{eff})

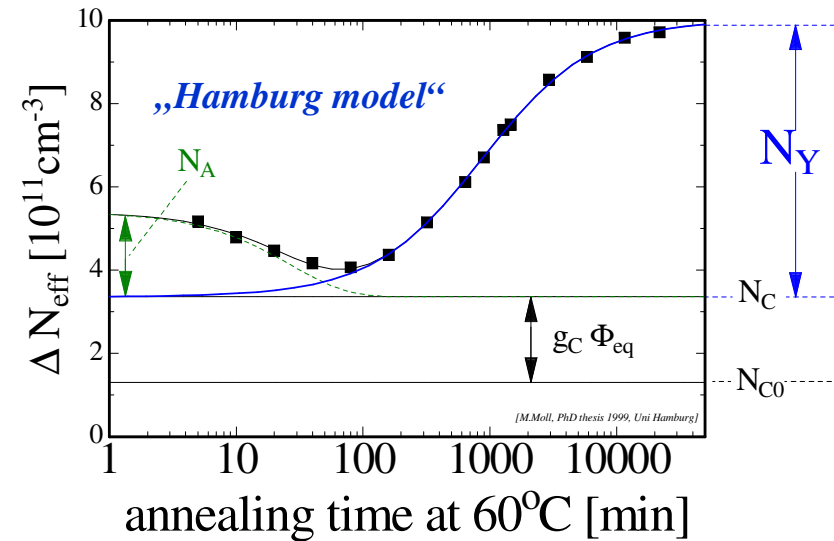
.... with particle fluence:



“Type inversion”: N_{eff} changes from positive to negative (Space Charge Sign Inversion)



.... with time (annealing):



Short term: **“Beneficial annealing”**

Long term: **“Reverse annealing”**

- time constant depends on temperature:

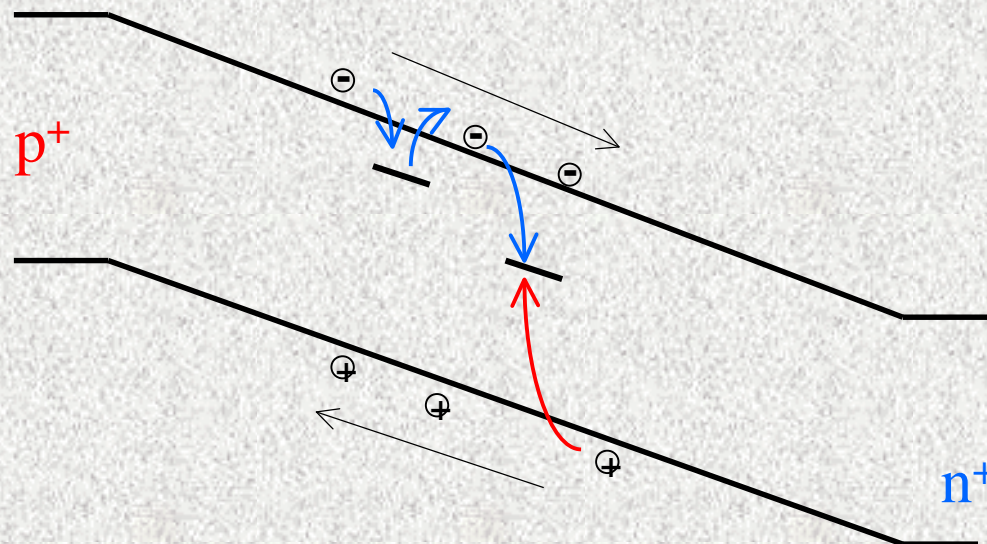
~ 500 years (−10°C)

~ 500 days (20°C)

~ 21 hours (60°C)

Consequence: Cool Detectors even during beam off (250 d/y)
alternative: acceptor/donor compensation by defect enginrg.

Charge trapping & recombination



- Increase of charge collection time
- Decrease of the charge collection efficiency
- Decrease of the S/N ratio

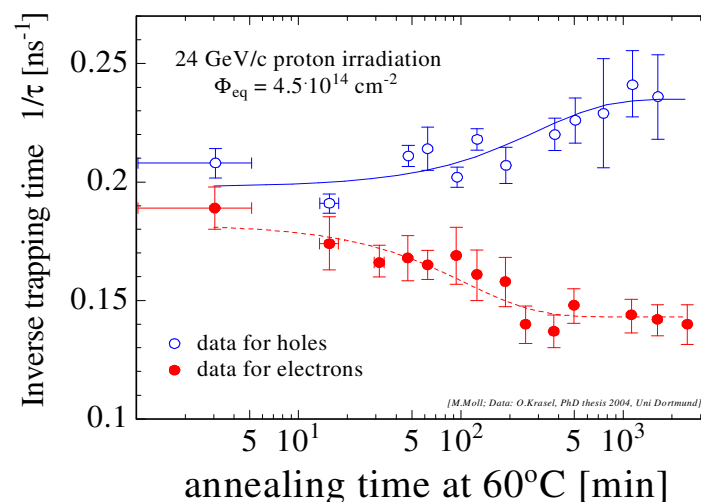
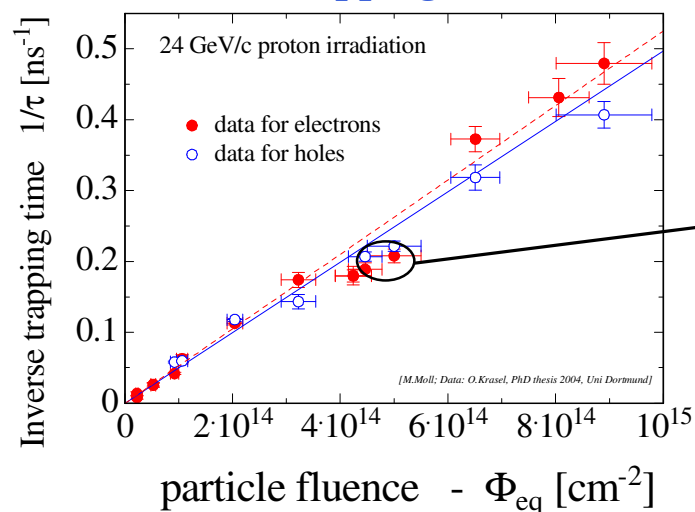
Radiation Damage – Charge carrier trapping

Deterioration of Charge Collection Efficiency (CCE) by trapping

Trapping is characterized by an effective trapping time τ_{eff} for electrons and holes:

$$Q_{e,h}(t) = Q_{0e,h} \exp\left(-\frac{1}{\tau_{\text{eff } e,h}} \cdot t\right) \quad \text{where:} \quad \frac{1}{\tau_{\text{eff } e,h}} \propto N_{\text{defects}}$$

Increase of inverse trapping time ($1/\tau$) with fluence and change with time (annealing):



Charge trapping leads to very small $\lambda_{e,h}$ at $\Phi_{\text{eq}} = 10^{16}/\text{cm}^2$

Consequence: Cooling does not help but: use thin detectors ($\sim 100\mu\text{m}$) and p-type Si

„Executive Summary“

- ❑ *Si-Detectors in the inner tracking area of future colliding beam experiments have to tolerate hadronic fluences up to $\Phi_{eq} = 10^{16}/\text{cm}^2$*
- ❑ *Deterioration of the detector performance is largely due to bulk damage caused by non ionizing energy loss (NIEL) of the particles*
- ❑ *Reverse current increase (most likely due to both point defects and clusters) is effectively reduced by cooling. Defect engineering so far not successful*
- ❑ *Change of depletion voltage is severe, also affected by type inversion and annealing effects. Modification by defect engineering is possible, for standard devices continuous cooling is essential (,freezing‘ of annealing)*
- ❑ *Charge trapping is possibly the ultimate limitation for Si-detectors, responsible defects are unknown, cooling and annealing have minor effects*

Outline



- ⌘ Some fundamentals about materials, device physics and processing related to Si detectors
- ⌘ Radiation tolerance of Si detectors, challenges and limiting factors
- ⌘ **Defect/impurity engineering of Si detectors**
- ⌘ New detector structures

The RD50 Collaboration



RD50: Development of Radiation Hard Semiconductor Devices for High Luminosity Colliders [\(http://rd50.web.cern.ch/rd50/\)](http://rd50.web.cern.ch/rd50/)

1. Formed in November 2001
2. Approved by CERN in June 2002

Main objective:

Development of ultra-radiation hard semiconductor detectors for the luminosity upgrade of LHC to $10^{35} \text{ cm}^{-2}\text{s}^{-1}$ (S-LHC)

Challenges:

- Radiation hardness up to fluences of 10^{16} cm^{-2} required;
- Fast signal collection (10 ns);
- Cost effectiveness.

Presently 280 Members from 55 Institutes

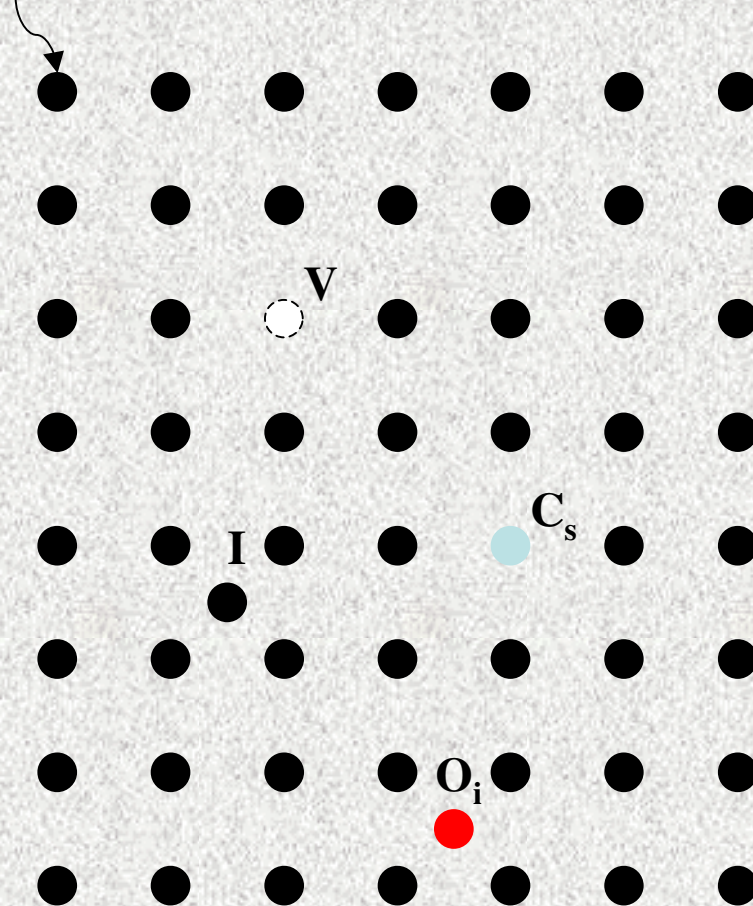
Belgium (Louvain), Canada (Montreal), Czech Republic (Prague (2x)), Finland (Helsinki (2x), Oulu), Germany (Berlin, Dortmund, Erfurt, Halle, Hamburg, Karlsruhe), Greece (Athens), Israel (Tel Aviv), Italy (Bari, Bologna, Florence, Milano, Modena, Padova, Perugia, Pisa, Trento, Trieste, Turin), Lithuania (Vilnius), Norway (Oslo (2x)), Poland (Warsaw), Romania (Bucharest (2x)), Russia (Moscow (2x), St.Petersburg), Slovenia (Ljubljana), Spain (Barcelona, Valencia), Sweden (Lund) Switzerland (CERN, PSI), Ukraine (Kiev), United Kingdom (Exeter, Glasgow, Lancaster, Liverpool, London, Sheffield, University of Surrey), USA (Fermilab, Purdue University, Rutgers University, Syracuse University, BNL, University of New Mexico)

Radiation effects in silicon detectors



Damage of the crystal lattice

Si atom (Si_s)



Electronic levels of main
electron traps in Si:

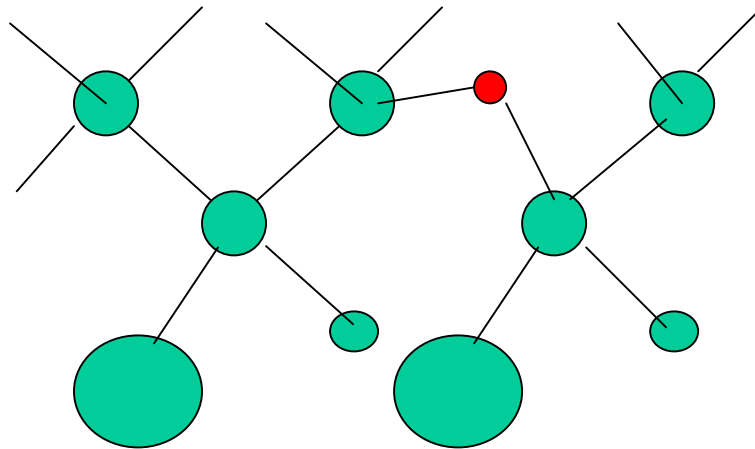
$$\begin{array}{c}
 E_c \quad \overline{\text{VO} \quad \text{V}_2} \\
 E_c - 0.18 \text{ eV} \quad \overline{-/0} \quad \overline{=/-} \quad E_c - 0.23 \text{ eV} \\
 \overline{-/0} \quad E_c - 0.44 \text{ eV} \\
 E_v \quad \overline{0/+} \quad E_v + 0.20 \text{ eV}
 \end{array}$$



Main types of Si-detector materials used

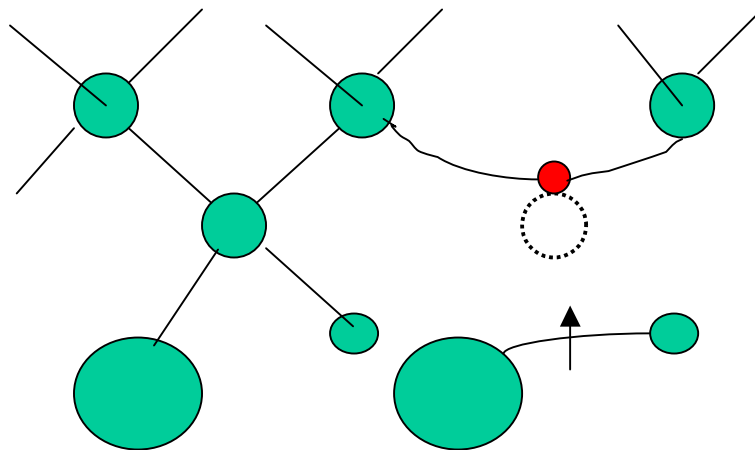
- Float zone (**Fz**); low oxygen and carbon content, $[O_i] < 5 \times 10^{15} \text{ cm}^{-3}$ and $[C_s] < 5 \times 10^{15} \text{ cm}^{-3}$
- Diffusion oxygenated Fz (**DOFZ**), $[O_i] \sim 3 \times 10^{17} \text{ cm}^{-3}$ and $[C_s] < 5 \times 10^{15} \text{ cm}^{-3}$
- Magnetic Czochralski (**MCz**), $[O_i] \sim 7 \times 10^{17} \text{ cm}^{-3}$ and $[C_s] < 5 \times 10^{15} \text{ cm}^{-3}$
- Epitaxial layers (**Epi**) on highly doped substrates, $[O_i] < 5 \times 10^{15} \text{ cm}^{-3}$ and $[C_s] < 5 \times 10^{15} \text{ cm}^{-3}$

Impurity engineering of high-purity Si



● Silicon

● Oxygen (O_i)
Interstitial configuration



Vacancy oxygen (VO)
center ($0/-$)

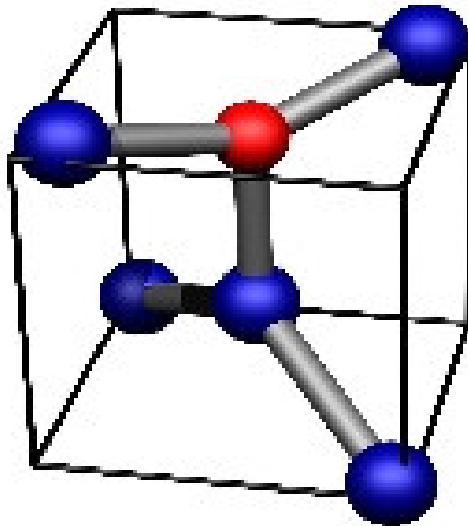
E_C ————— } 0.18 eV
 ↑

E_V —————

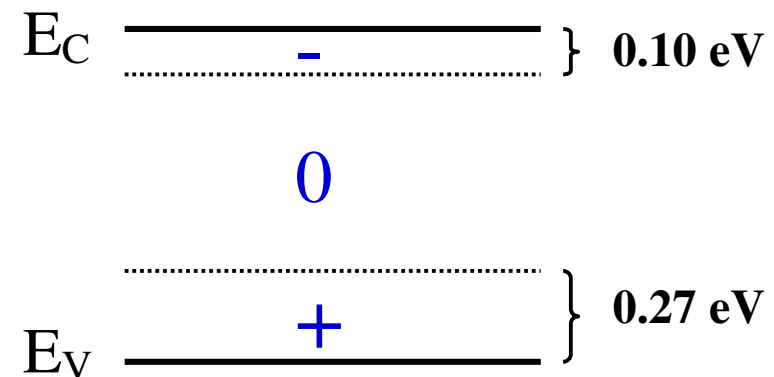


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Interstitial carbon (C_i)



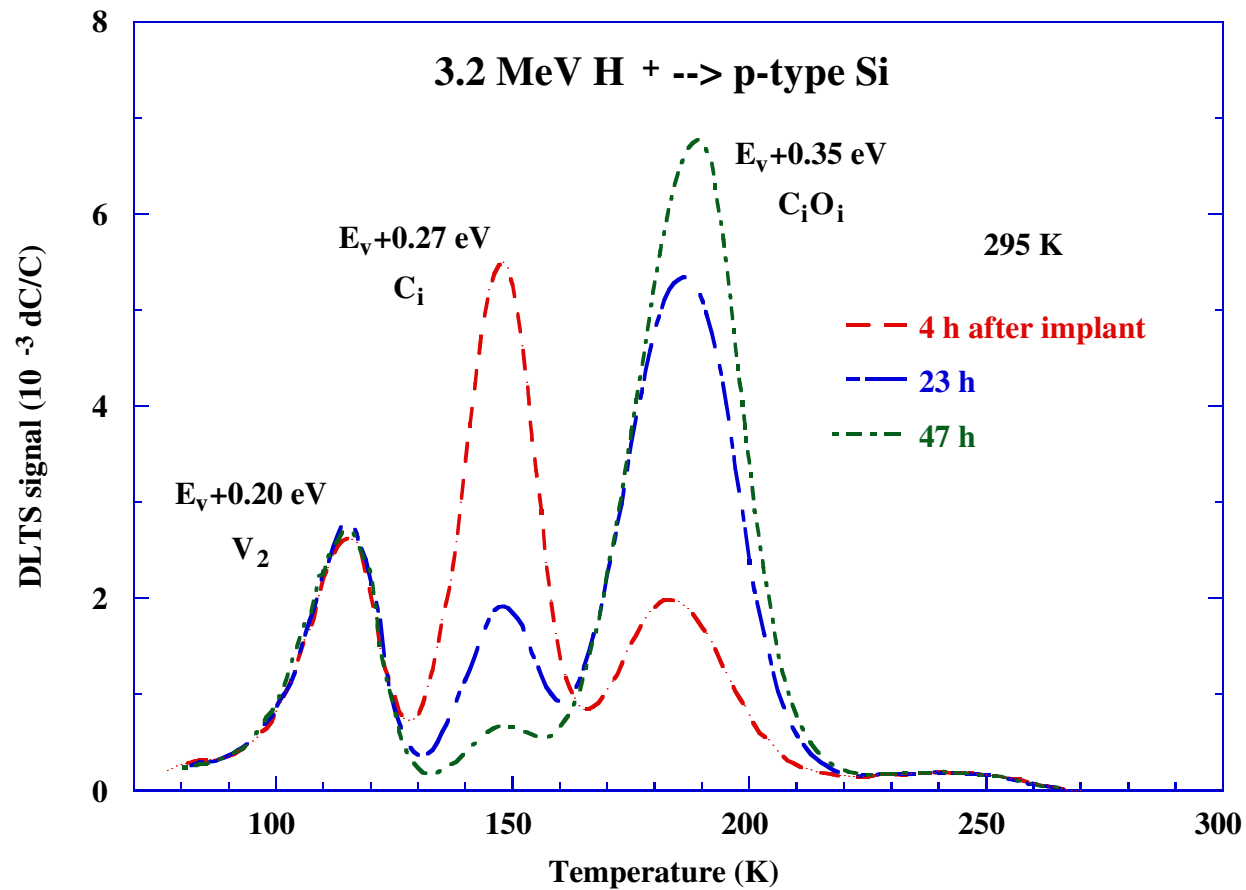
Three different charge states (-,0,+)



C_s has a strong impact on the overall defect generation via its role as I-trap; $C_s + I \rightarrow C_i$

Carbon is of key importance in n^-/p^- detector layers, either directly or indirectly

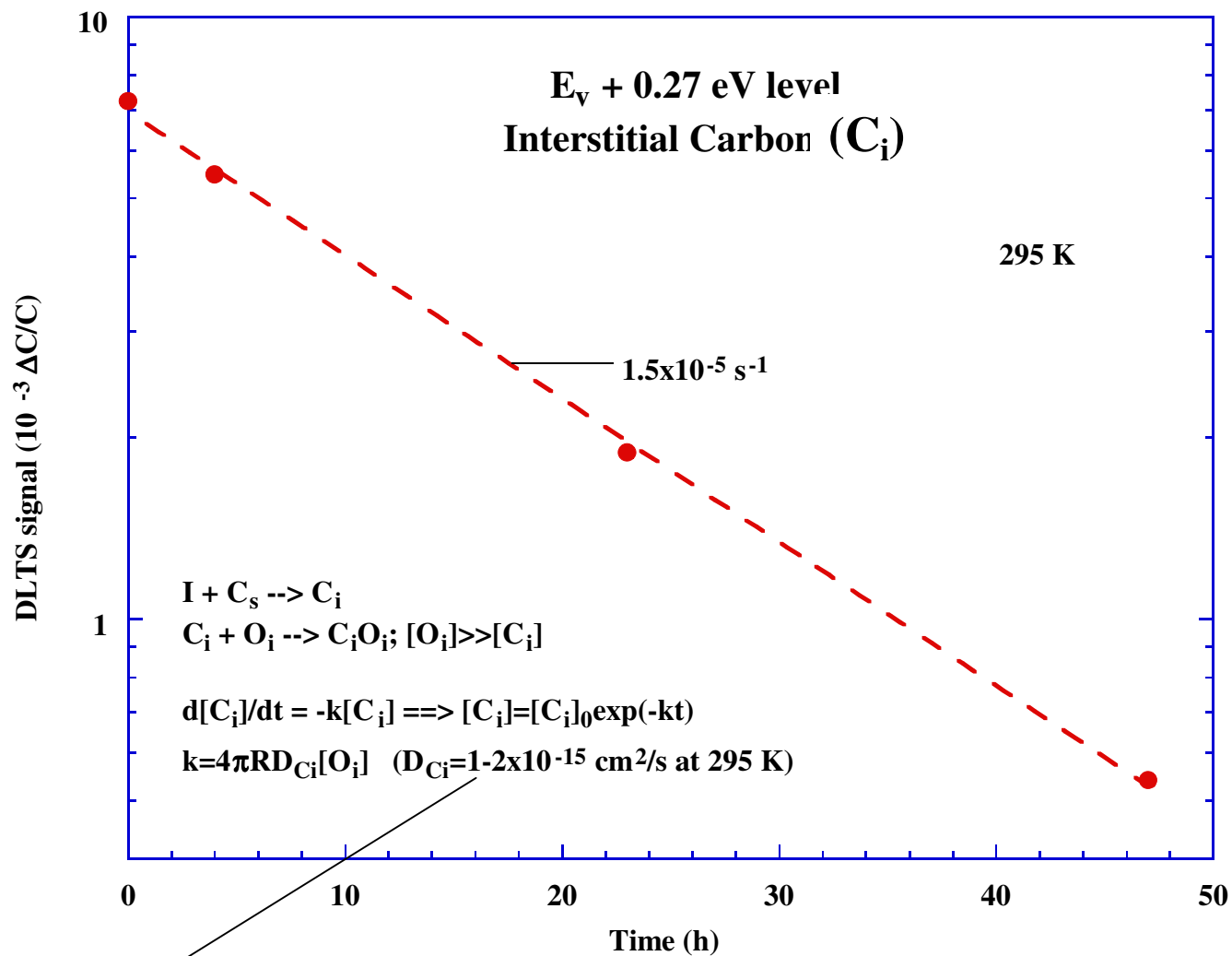
Evolution of C_i at RT; an illustration



Fz ($\sim 15 \Omega\text{cm}$)

$[O_i] \sim 1.2 \times 10^{16} \text{ cm}^{-3}$

$[C_s] \leq 5 \times 10^{15} \text{ cm}^{-3}$



Cf value by Tipping, Newman, Semicond. Sci. Techn. **2**, 315 (1987)
 $D_{Ci} = 0.4 \exp(-0.87(\text{eV})/kT) \text{ cm}^2/\text{s}$

Lalita et al., NIMB **120**, 27 (1996)

Purpose of defect engineering



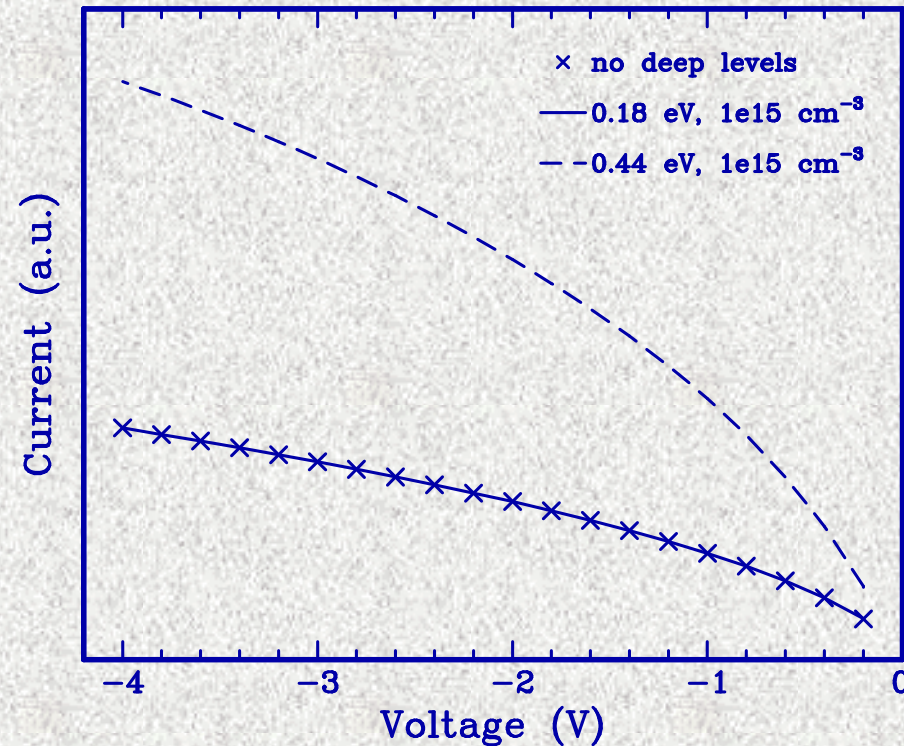
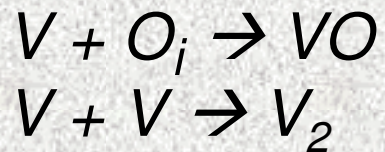
$$E_c \quad \overline{VO \quad V_2}$$

$$E_c - 0.18 \text{ eV} \quad \overline{-/0} \quad \overline{\equiv/-} \quad E_c - 0.23 \text{ eV}$$

$$\overline{-/0} \quad E_c - 0.44 \text{ eV}$$

$$\overline{0/+} \quad E_v + 0.20 \text{ eV}$$

$$E_v \quad \overline{\quad \quad \quad}$$

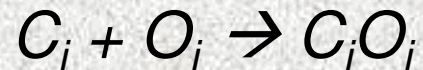
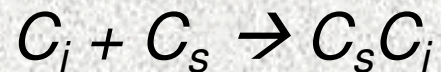
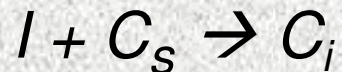
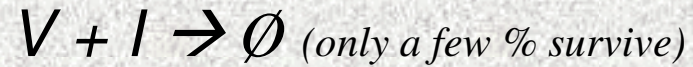


Leakage current as a function of the bias in silicon detector with different types of defects (SILVACO TCAD).

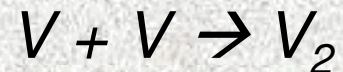
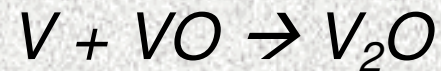
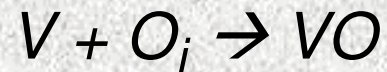
Monakhov et al., Sol. St. Phen. 82-84, 441 (2002)

Suppress formation of defects with levels close to mid-gap !!

Key defect reactions and why oxygenation



.....



C_s traps ('immobilizes') I and suppresses self-annihilation ☹️

O_i traps V and suppresses V-clustering 😊

(Standard interpretation of oxygenation effect....)

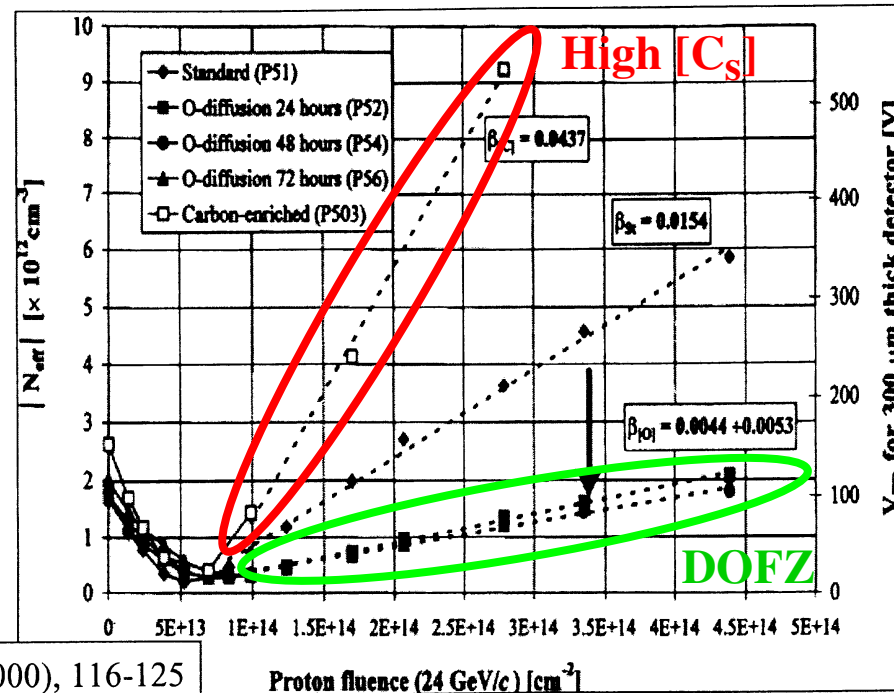
Impurity engineering of high-purity Si

DOFZ Si is significantly radiation harder than standard Si for
 γ , π , p irradiations
 Almost no effect for neutron irradiations

$$N_{eff} = |N_c(0) \cdot e^{-c\phi} - \beta \cdot \phi|$$

$$\beta_{\text{standard Si}} \sim 3 \beta_{\text{DOFZ}}$$

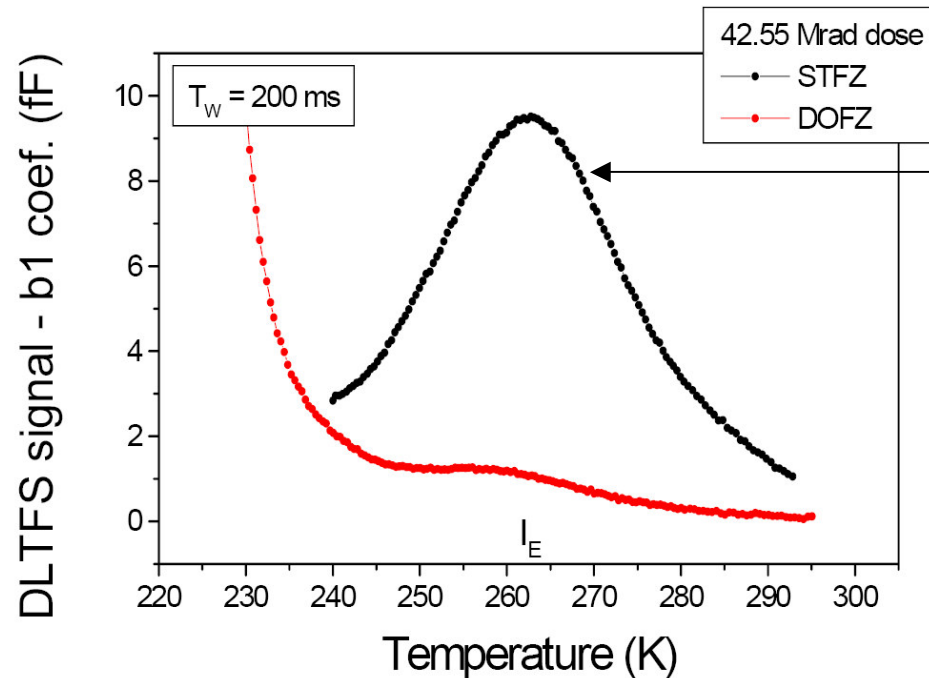
Reverse annealing
 significantly reduced



RD48, NIM A 447 (2000), 116-125

ATLAS-Pixel collaboration has now adopted DOFZ Si
 CMS-Pixel is considering this option

γ -irradiated Si-detectors



I-center

$E_c - 0.545$ eV

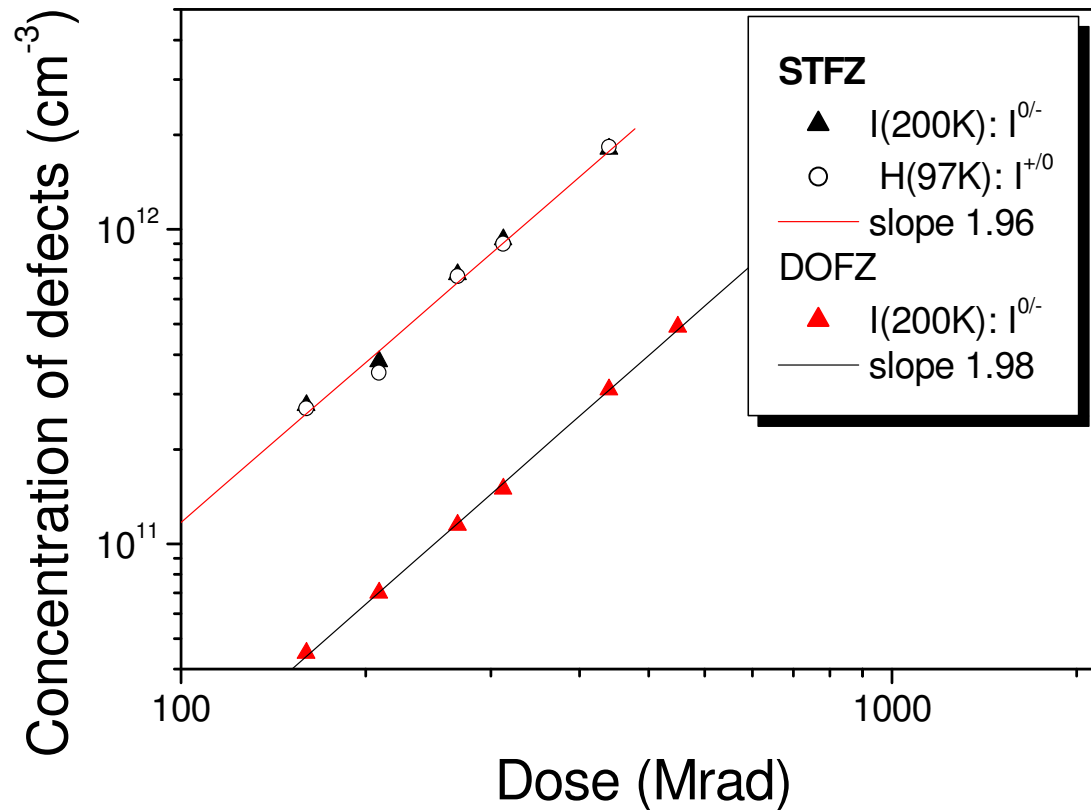
$\sigma_n = 1.7 \times 10^{-15}$ cm²

$\sigma_h = 9 \times 10^{-14}$ cm²

Key defect for space charge inversion,...

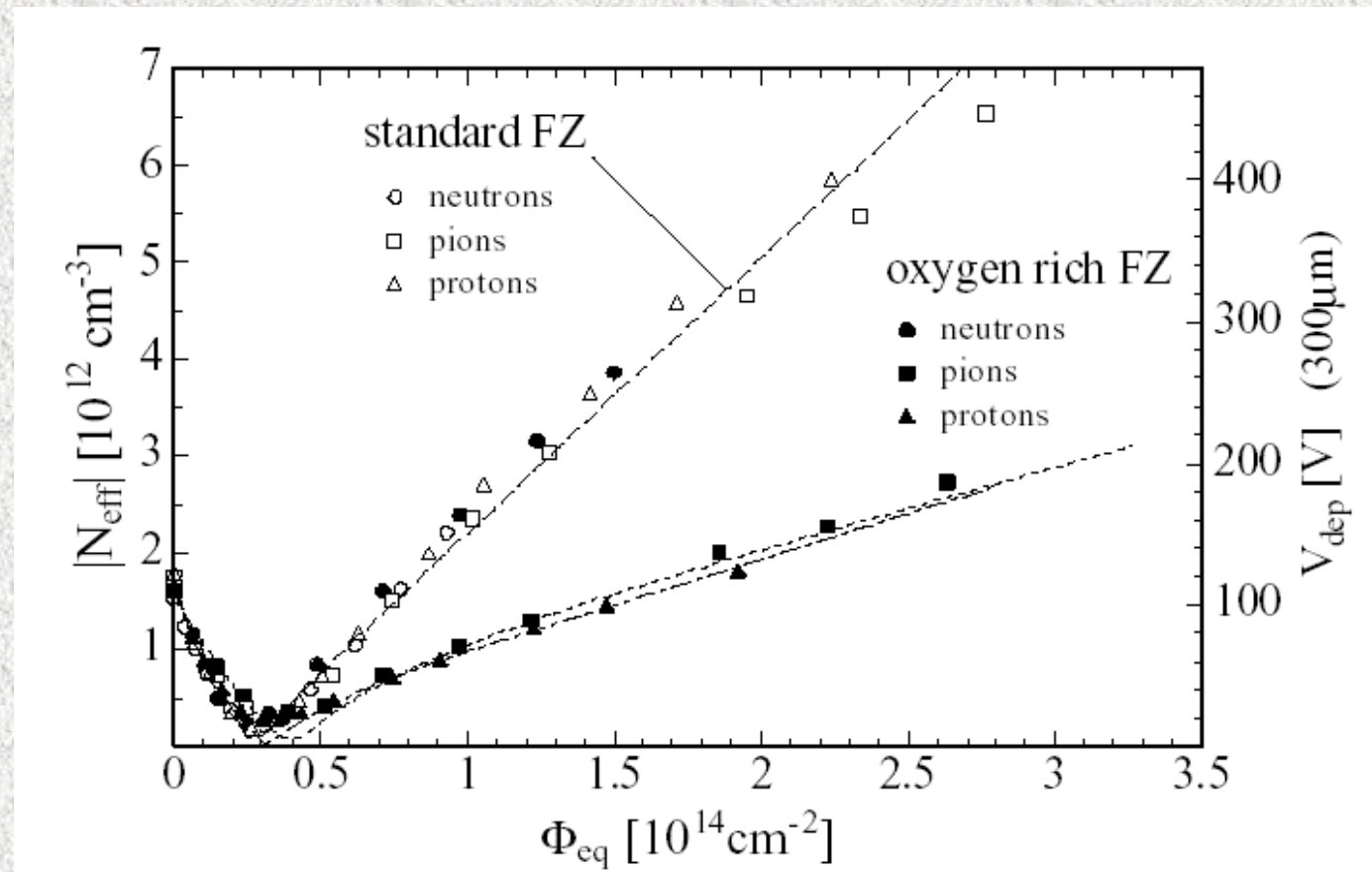
Identity of I??

γ -irradiated Si-detectors



The I-center has a quadratic dose dependence → 'simple' cluster-type defect

No radiation hardening for neutrons



Oxygenation does not affect direct cluster formation!?

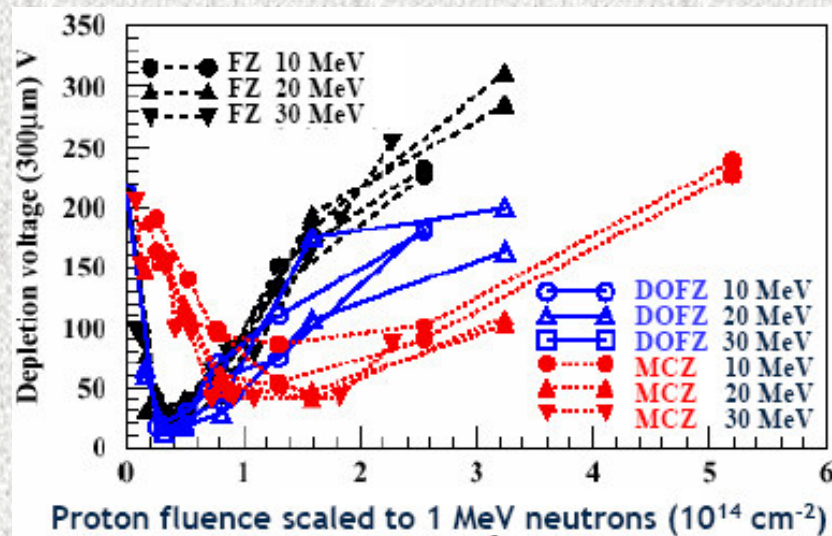
Improving Si radiation hardness



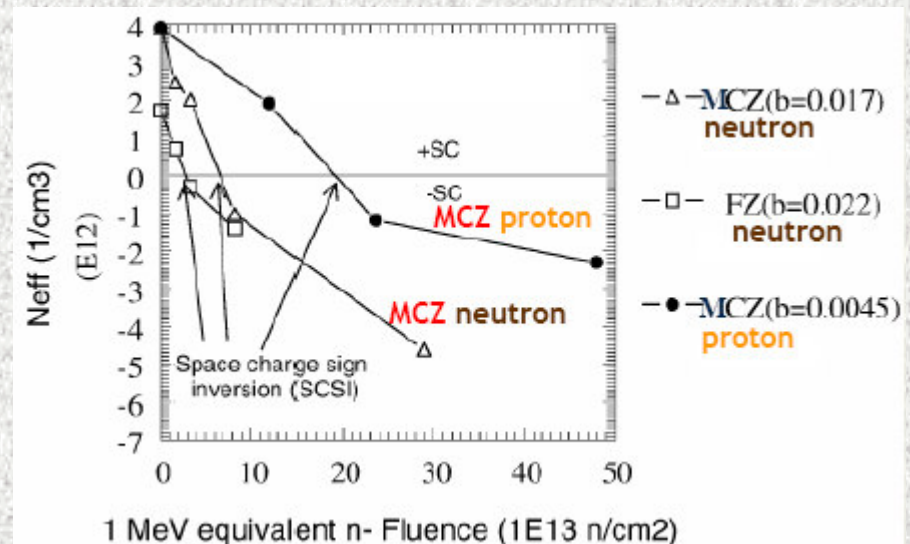
Magnetic Czochralski (MCZ) silicon

standar FZ-Si: $[O] \sim 10^{16} \text{ cm}^{-3}$
oxygenated FZ-Si: $[O] \sim 1\text{-}4 \times 10^{17} \text{ cm}^{-3}$
MCZ-Si: $[O] \sim 10^{18} \text{ cm}^{-3}$

Higher radiation tolerance...



...but the same 'neutron problem'.



Summary of defect/impurity engineering



- Oxygenation (shallow VO instead of deep levels)
- Decrease in carbon content (C effects vacancy-interstitial annihilation) → *argument for epi-detectors*
- Hydrogenation (H passivates dangling Si bonds)
- Formation of electrically inactive extended defects (sink for vacancies and interstitials)
- Engineering of 'directly created' clusters (neutron irradiation) is a challenge...

Outline



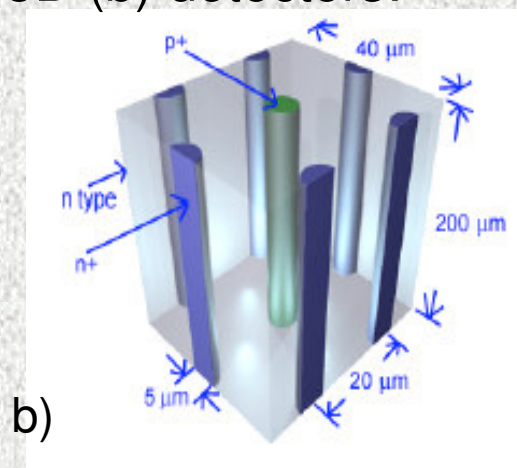
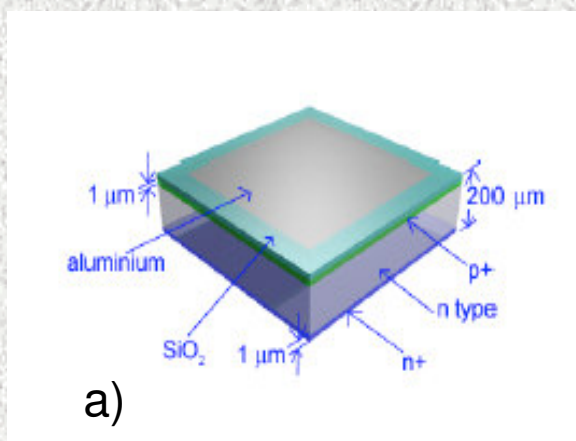
- ⌘ Some fundamentals about materials, device physics and processing related to Si detectors
- ⌘ Radiation tolerance of Si detectors, challenges and limiting factors
- ⌘ Defect/impurity engineering of Si detectors
- ⌘ **New detector structures**

3D detectors



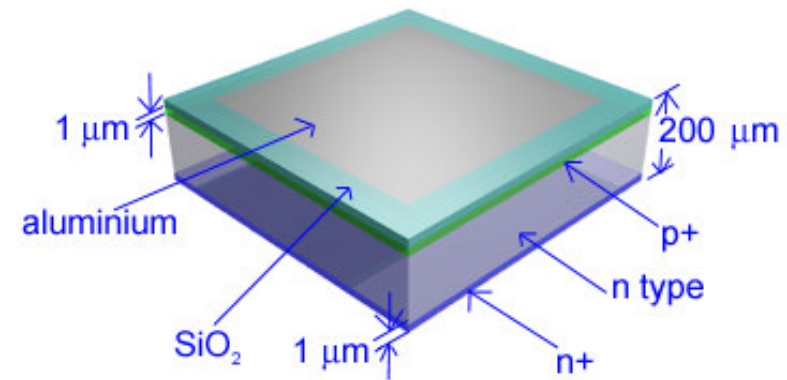
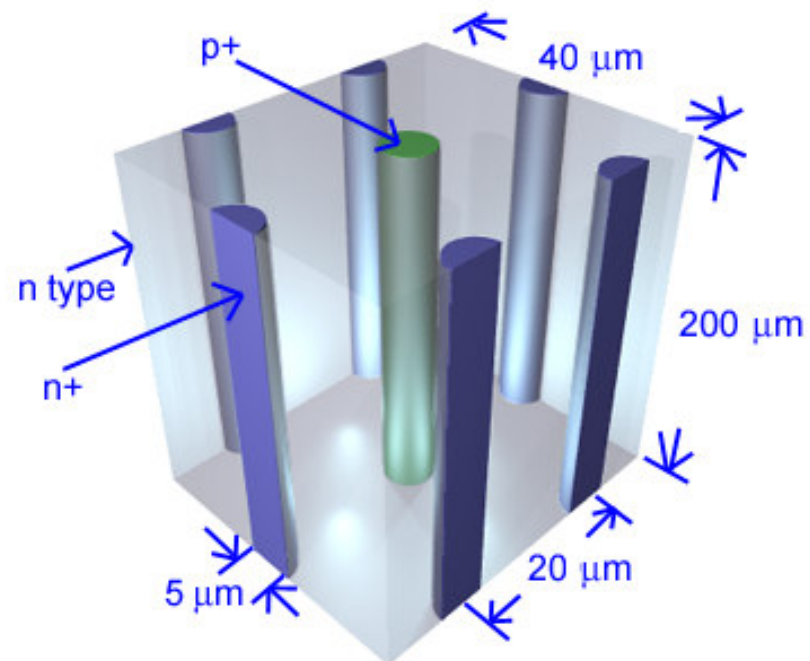
- Proposed by [S.I. Parker, C.J. Kenney and J. Segal](#) (NIM A 395 (1997) 328)
- Called 3-D because, in contrast to silicon planar technology, have **three dimensional (3-D) electrodes penetrating the silicon substrate**
- Presently, a joint effort exists between Brunel Univ., Hawaii Univ., Stanford Univ., SINTEF and UiO

Conventional (a) and 3D (b) detectors:



depletion thickness depends on **p+ and n+ electrode distance**, not on the substrate thickness → (1) can operate at **very low voltages** or (2) can have a high doping for **ultra-high radiation hardness**, and (3) short distance for charge collection

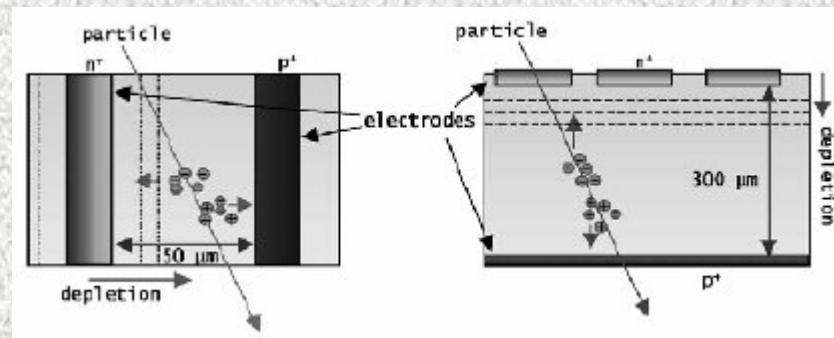
Schematics of 3D- and ordinary detector structures



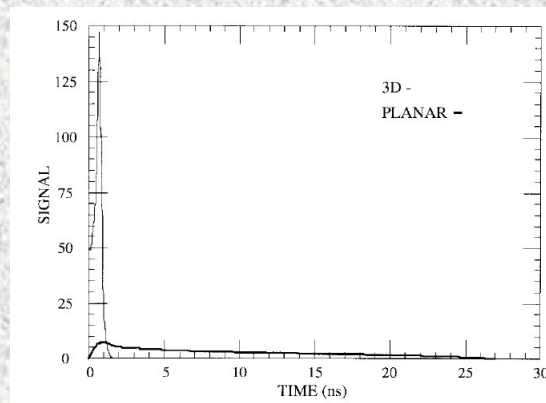
3D detectors



Charge collection in 3D detectors



- shorter **collection length** than planar technology
- shorter **charge collection time** than planar technology
- higher **charge collection efficiency**



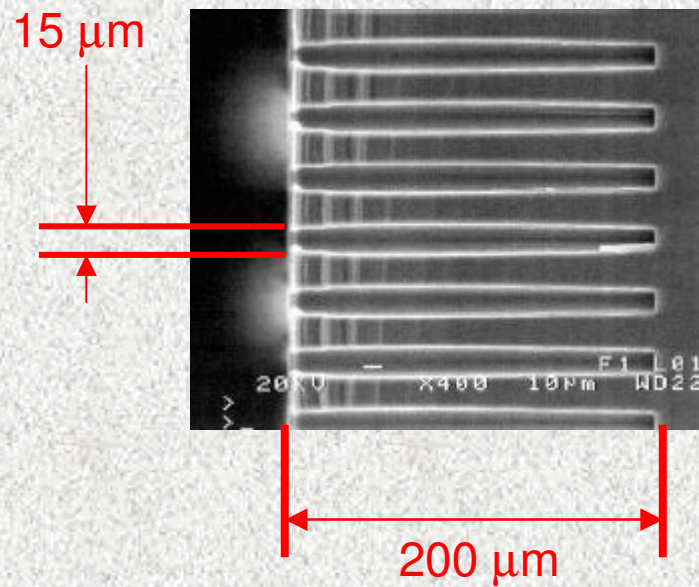
computer simulations of the charge collection dynamics for planar and 3D detectors

3D detectors

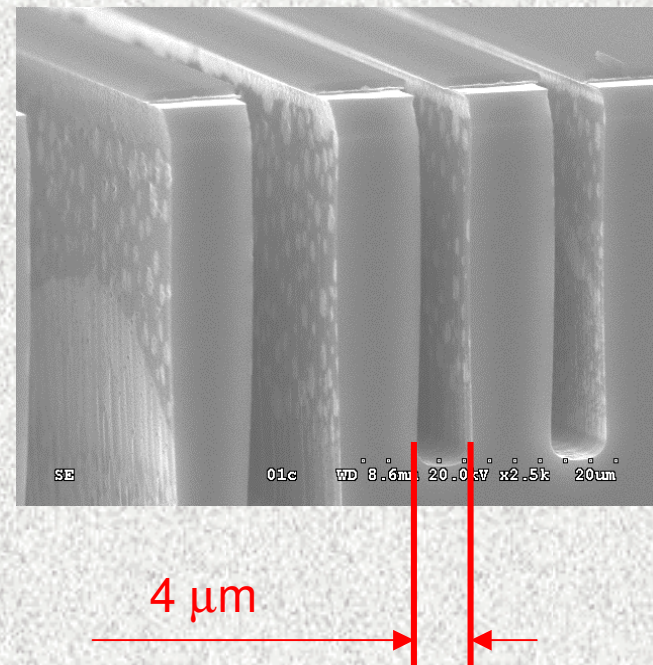


Real devices

a 3D detector structure:



a 3D structure etched at SINTEF:

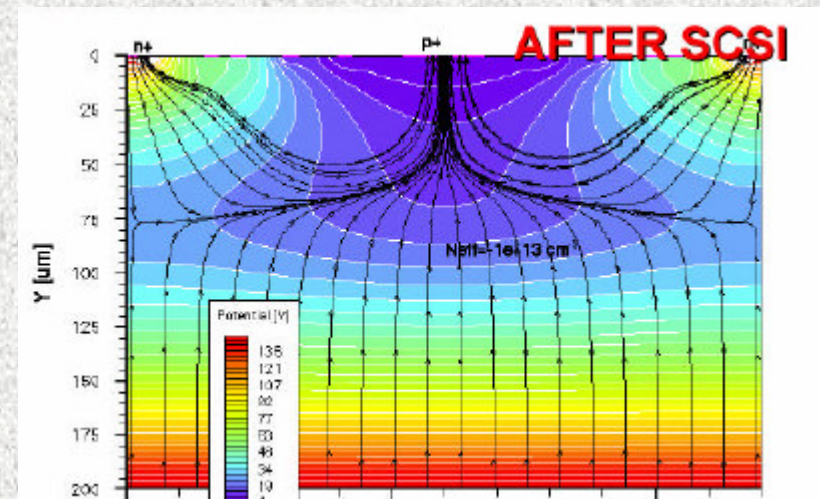
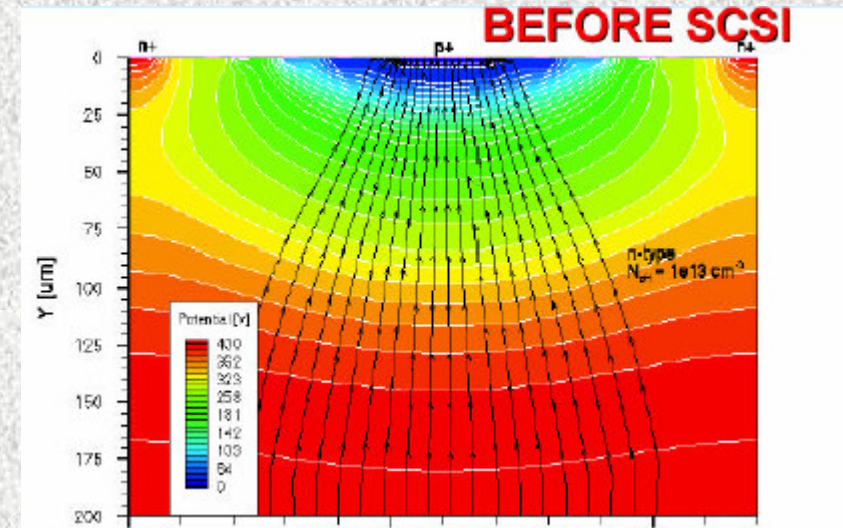
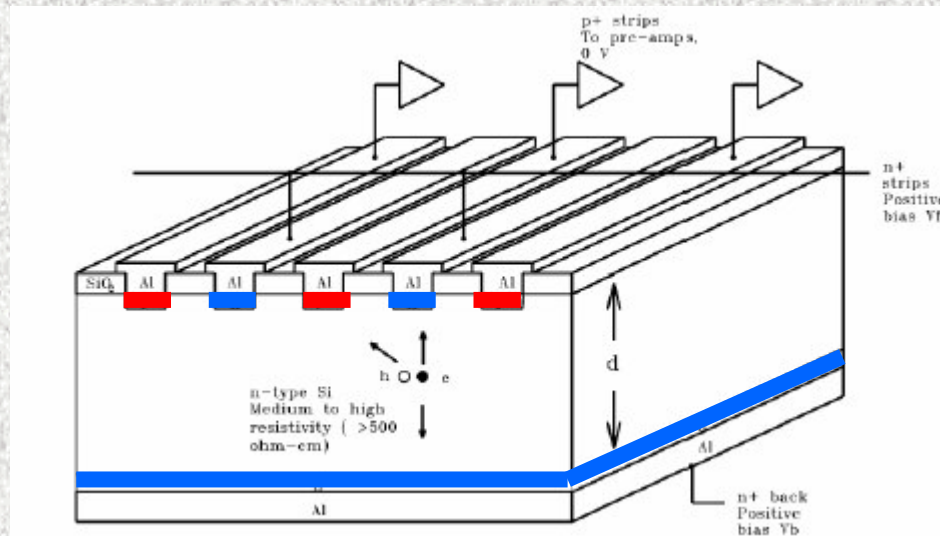


Semi-3D detectors



Proposed by Z. Li (NIM A 478 (2002) 303).

Single-side detectors with alternating p- and n- strips on the front side



After SCSi, the depletion occurs from both sides reducing the needed depletion voltage by factor 2.5

FINAL SUMMARY



- ✓ There is considerable activity and **progress** in improving radiation hardness of Si particle detectors, reaching 10^{16} cm^{-2} (1 MeV n eq.) is, indeed, a challenge
- ✓ Using the so-called **defect/impurity engineering**, the range of working fluences has been extended up to **$>10^{15} \text{ cm}^{-2}$**
- ✓ The progress in semiconductor **microtechnology** allows design of detector structures with 'inherent' **ultra-high radiation tolerance**; development of a viable industrial '3D technology' is in progress

Thin detectors



Advantages:

1. Smaller leakage current: $I_{\text{leak}} \propto W$
2. Smaller depletion voltage: $V_{\text{dep}} = qW^2N_{\text{eff}}/2\epsilon \propto W^2$
3. Lower probability for trapping and recombination ($\propto W$)

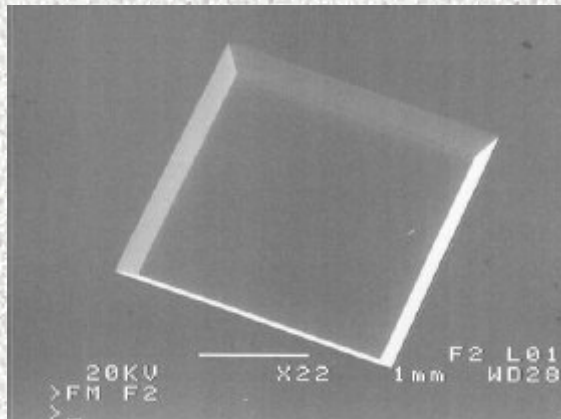
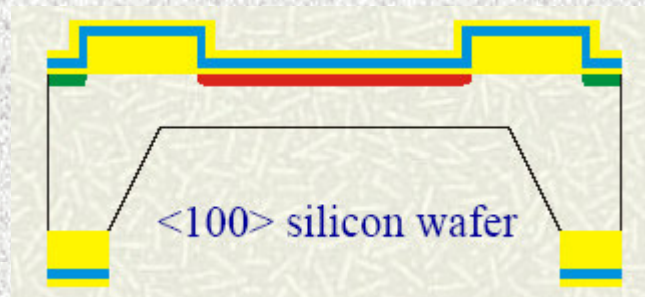
Disadvantage:

Smaller amount of carriers generated by a particle, i.e., smaller amplitude of the signal ($\propto W$)

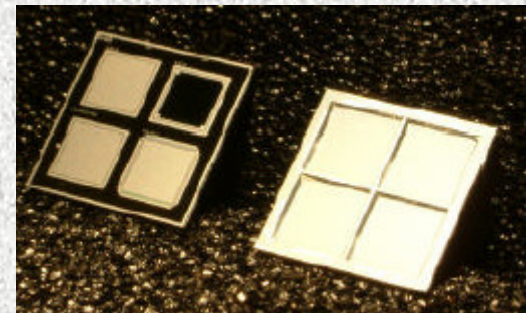
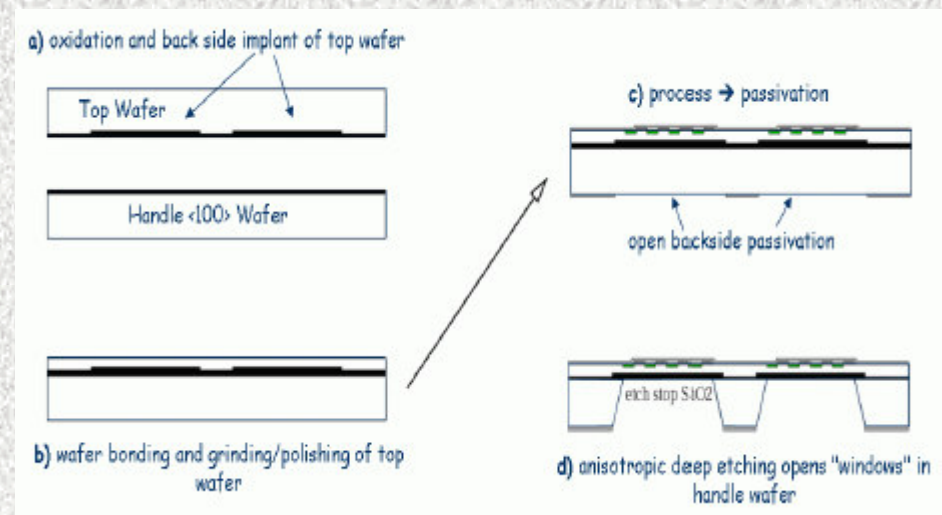
Thin detectors



FZ thinned devices ($W=50$ mm)



SEM: back view of a thinned device
Area: $1 \text{ mm}^2 - 20 \text{ mm}^2$ and
 $I < 1 \text{ nA/cm}^2$ at 20 V



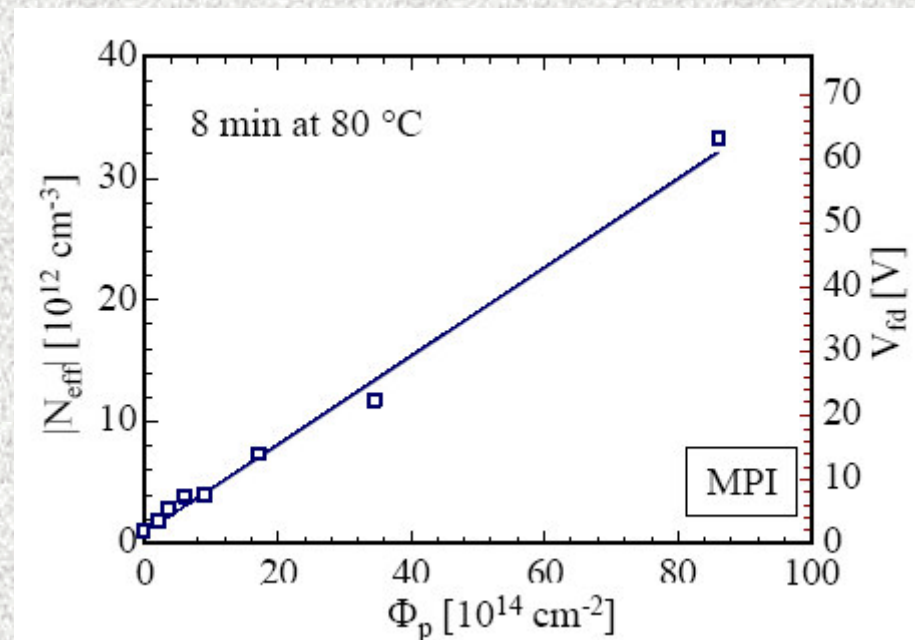
Front (left) and back (right) view of thinned devices
Area: 10 mm^2 and $I < 1 \text{ nA/cm}^2$ at 20 V

Thin detectors



FZ thinned devices (W=50 mm): Radiation hardness

20 GeV proton irradiation: $\Phi_p = 9.5 \times 10^{13} - 8.6 \times 10^{15} \text{ cm}^{-2}$

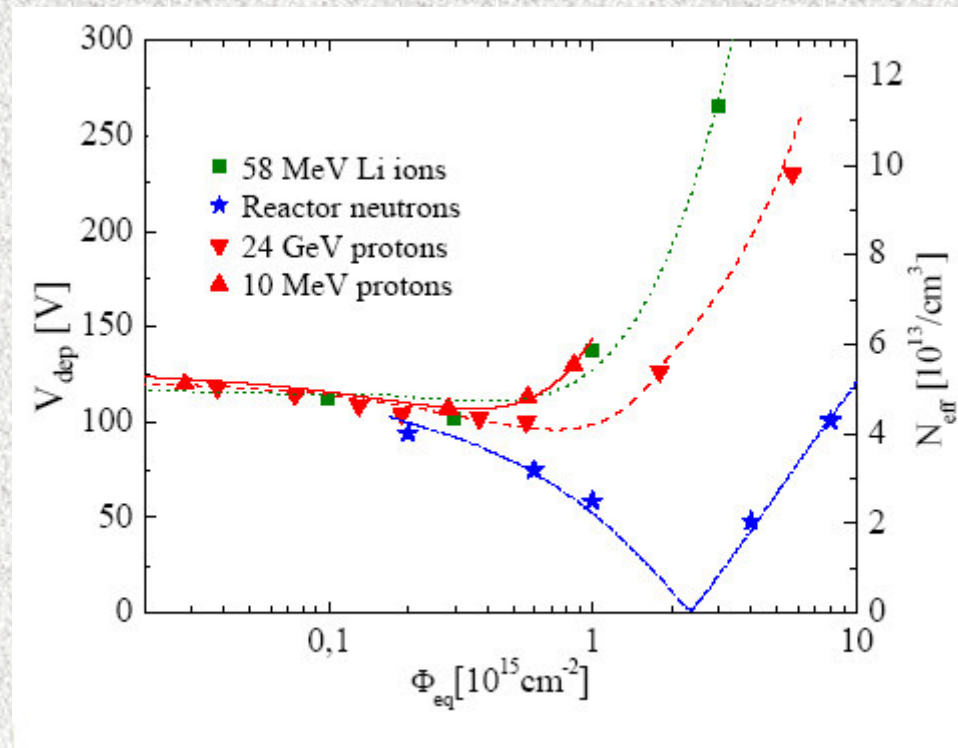
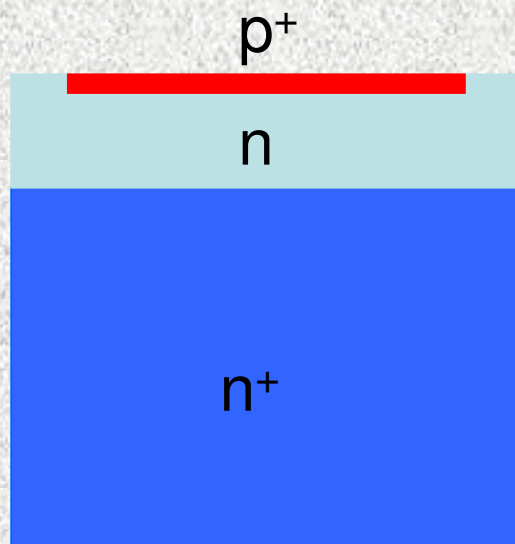


for $W=300 \text{ }\mu\text{m}$ at $\Phi=8.6 \times 10^{15} \text{ cm}^{-2}$: $V_{\text{dep}} \sim 2300 \text{ V}$

Thin detectors



Epitaxially grown layer ($W=50\text{ }\mu\text{m}$)

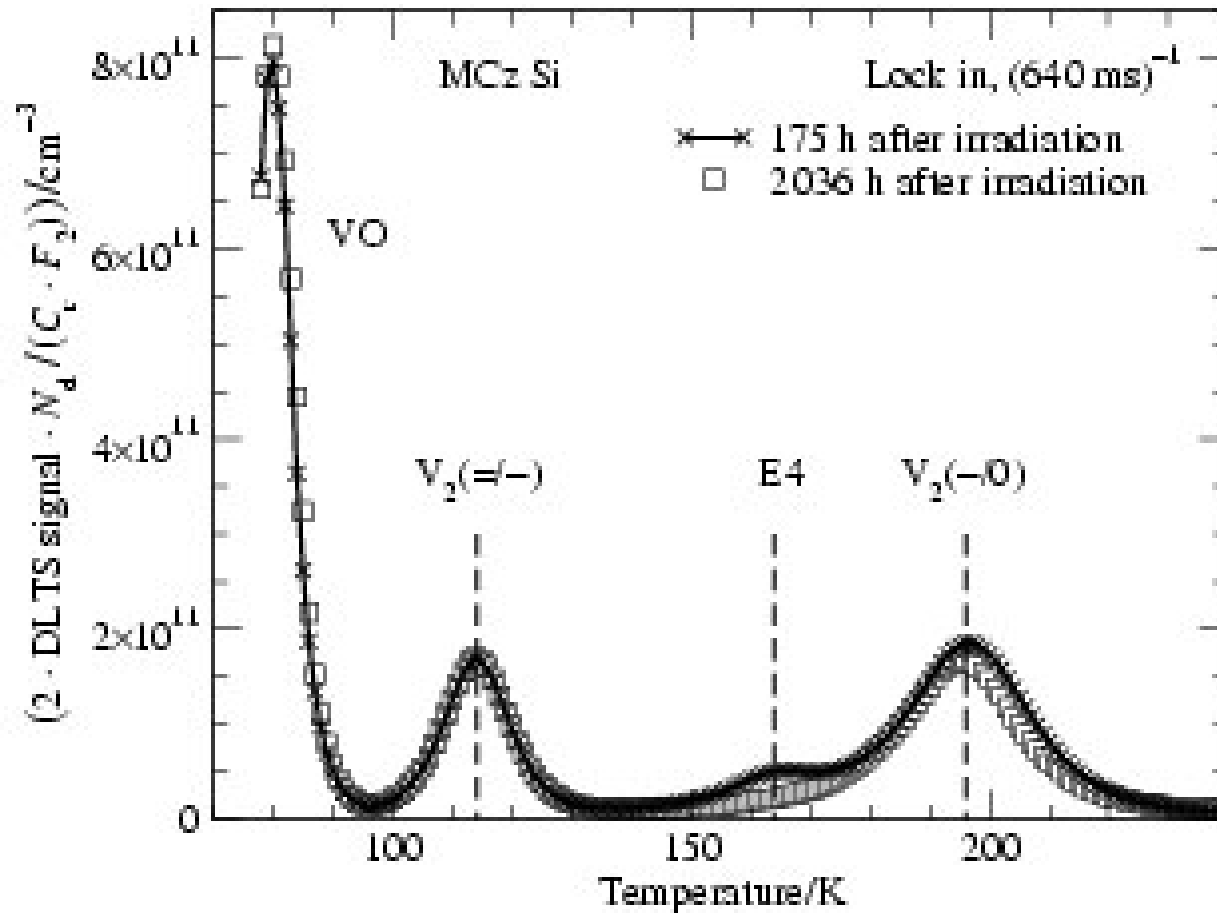


The smaller thickness of the detecting region (epi-layer) allows to increase the doping and still use relatively low bias ($W \propto \sqrt{V/N_d}$)
Higher doping shifts the SCSI point to higher fluences.



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Typical DLTS-spectra



P⁺-n⁻-n⁺ MCz diode

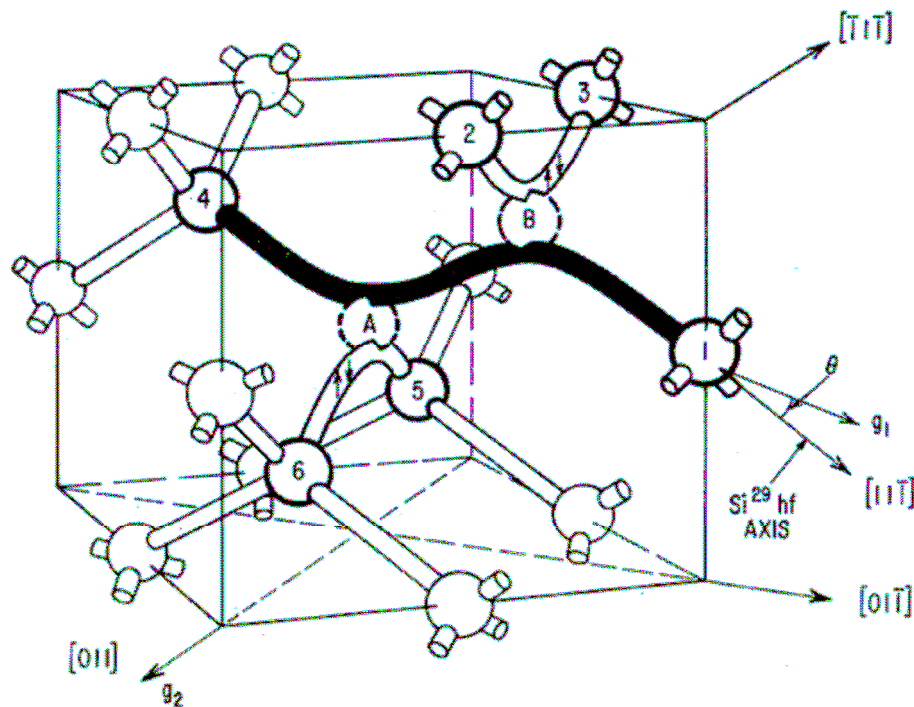
$N_d \sim 5 \times 10^{12} \text{ cm}^{-3}$

6 MeV e⁻, $5 \times 10^{12} \text{ cm}^{-2}$



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The V_2 -center



Four different charge states ($=, -, 0, +$)
with corresponding levels at $E_c - 0.23$,
 $E_c - 0.43$ and $E_v + 0.20$ eV

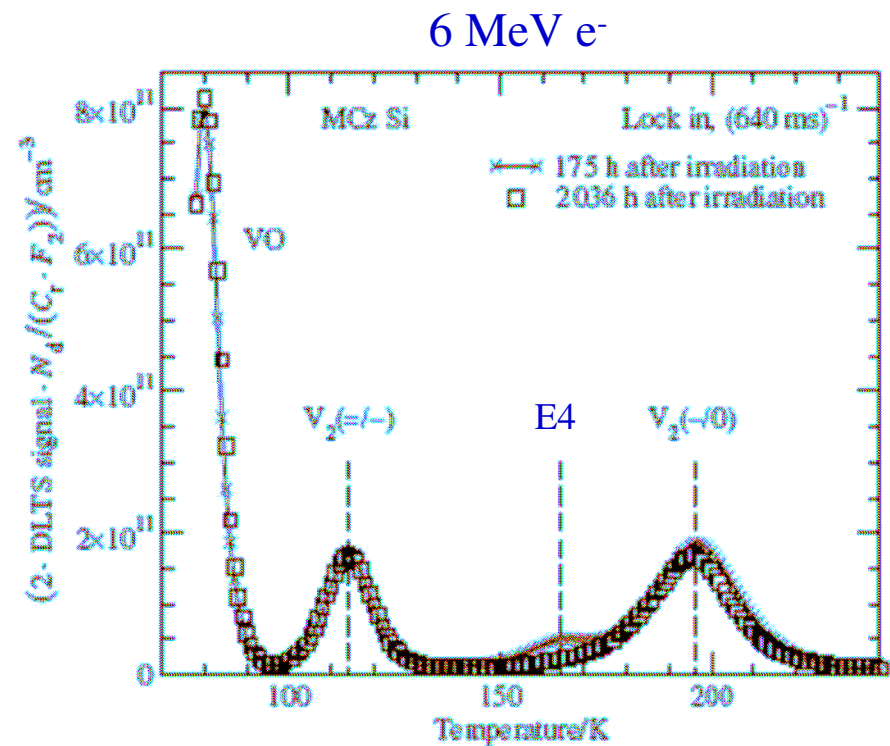
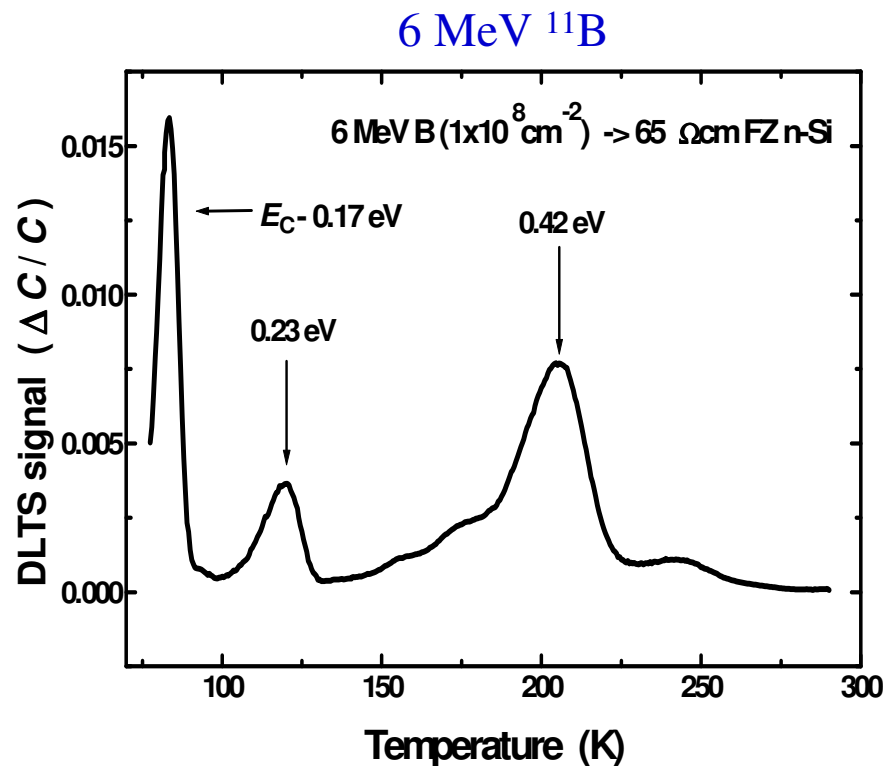
The most prominent intrinsic defect
stable at RT

J.W. Corbett, G.D. Watkins, Phys. Rev. Lett. **7**, 314 (1961)

.....

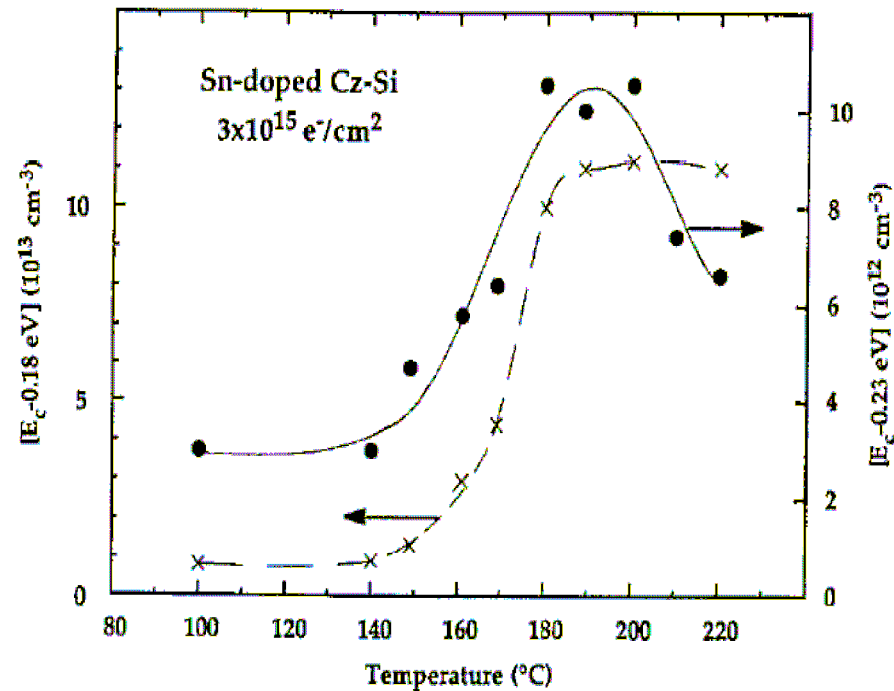
Presumably of key importance in n^-/p^- detector layers,
either directly or indirectly

Generation of VO and V_2 ; *mass effect*

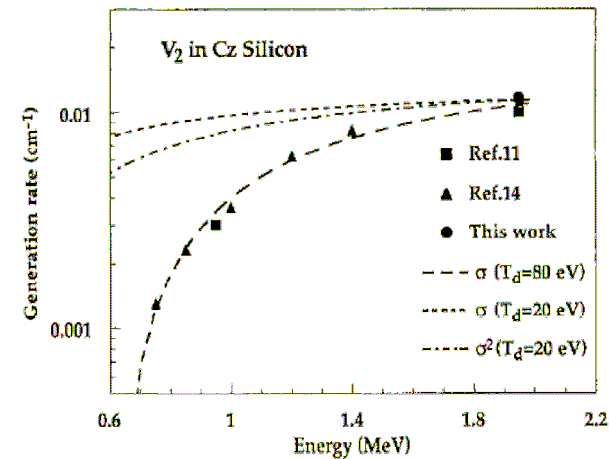
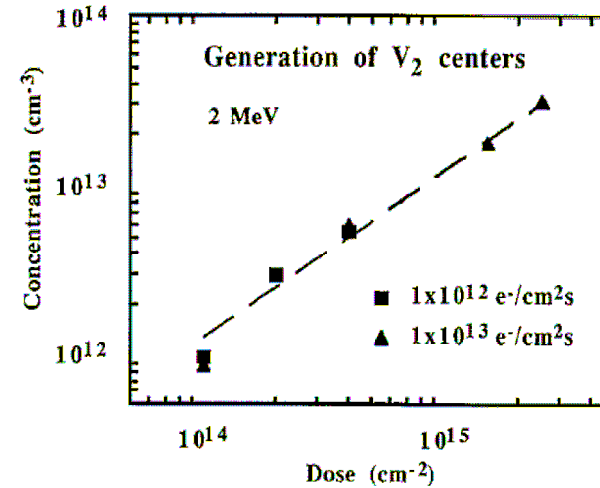


Importance of V_2 (and higher order clusters) increases with increasing elastic energy deposition (NIEL), e.g., neutral hadrons

Generation mechanism for V_2



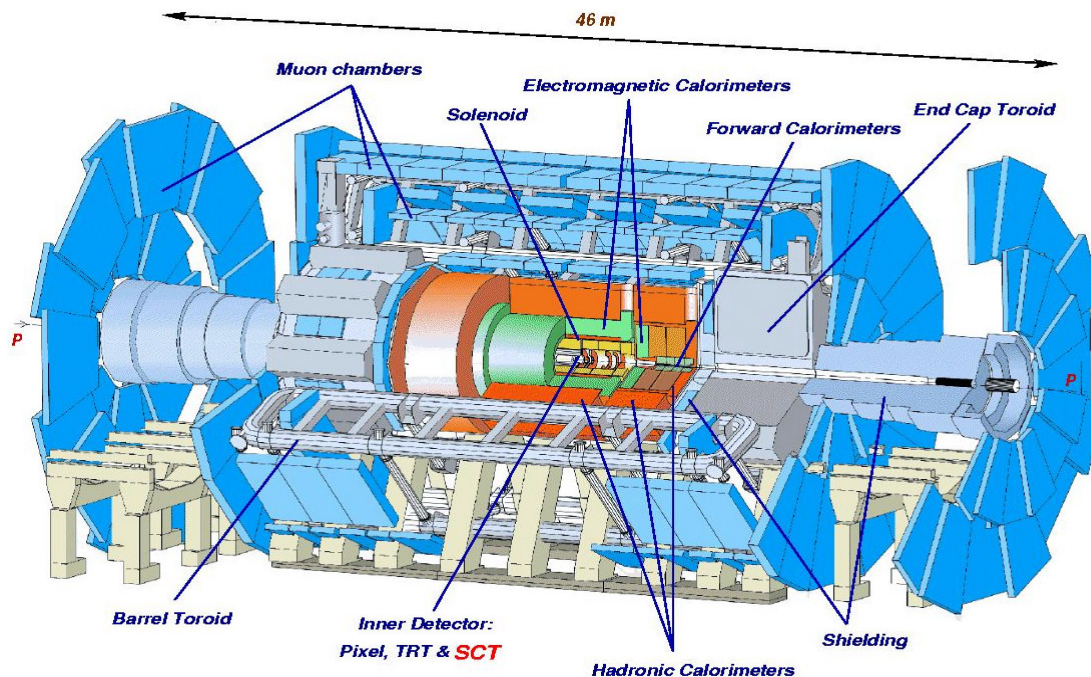
Svensson, Lindström, J. Appl. Phys. **72**, 5616 (1992)



'Direct' generation of V_2 prevails (pairing of V's formed by different impinging electrons is negligible)!

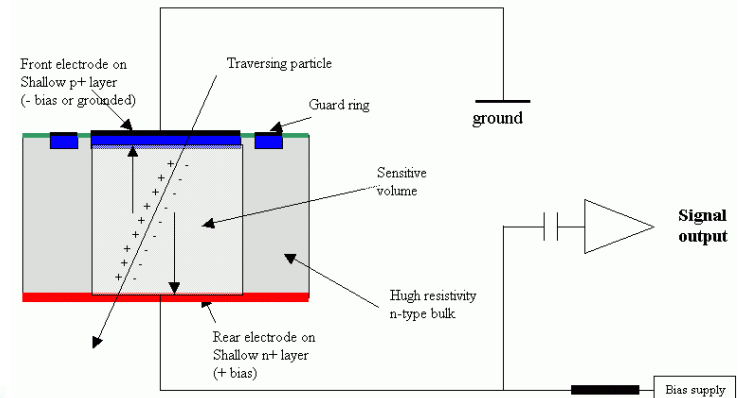
LHC ATLAS Detector – a Future HEP Experiment

Overall length: 46m, diameter: 22m,
total weight: 7000t, magnetic field: 2T
ATLAS collaboration: 1500 members

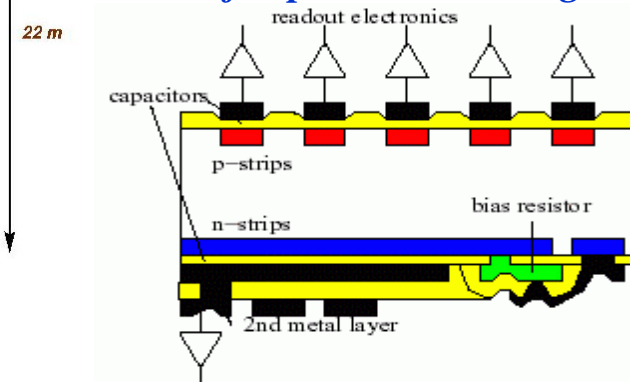


2nd general purpose experiment:
CMS, with all silicon tracker!

*principle of a silicon detector:
solid state ionization chamber*



*micro-strip detector
for particle tracking*



For innermost layers: pixel detectors

Acknowledgements



Financial support from

- the Norwegian Research Council (NFR - Strategic programs on micro/nanotechnology and materials science (NANO/FUNMAT))
- the Nordic Research Training Academy (NorFA)
- University of Oslo (Functional materials program)

is gratefully acknowledged.

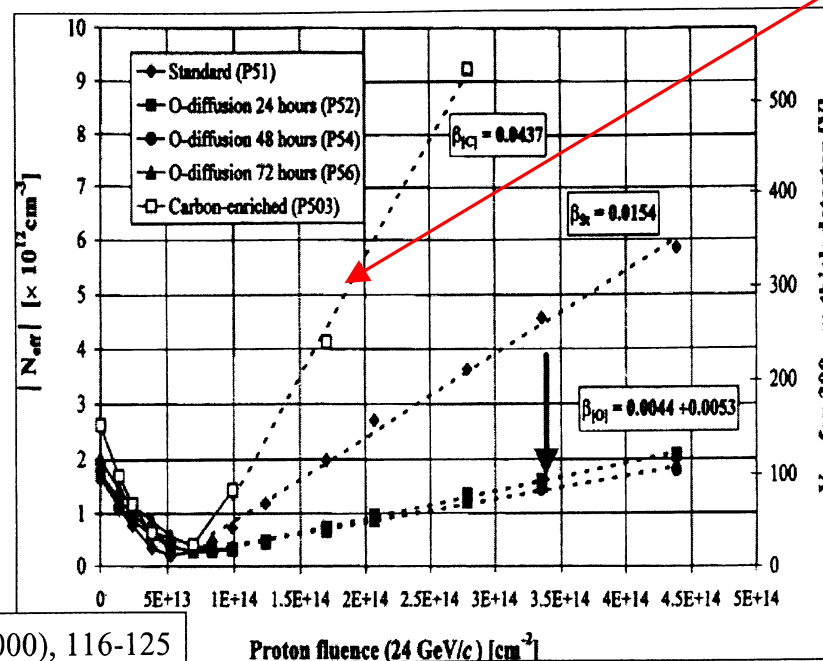
Impurity engineering of high-purity Si

DOFZ Si is significantly radiation harder than standard Si for γ, π, p irradiations
Almost no effect for neutron irradiations

$$N_{eff} = |N_c(0) \cdot e^{-c\phi} - \beta \cdot \phi|$$

$$\beta_{\text{standard Si}} \sim 3 \beta_{\text{DOFZ}}$$

Reverse annealing
significantly reduced



RD48, NIM A 447 (2000), 116-125

ATLAS-Pixel collaboration has now adopted DOFZ Si
CMS-Pixel is considering this option

The role of C_s may be indirect